# TIME05 – Workshop on Tracking In high Multiplicity Environments October 3-7, Zürich, Switzerland

# Radiation Tolerant Semiconductor Sensors for Tracking Detectors

### **Michael Moll**

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on behalf of the

- CERN-RD50 project –

http://www.cern.ch/rd50

# **Outline**



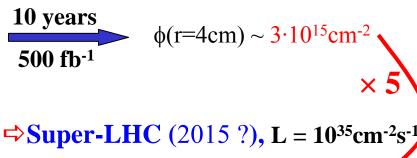
- Motivation to develop radiation harder detectors: Super-LHC
- Introduction to the RD50 collaboration
- Radiation Damage in Silicon Detectors (A review in 5 slides)
  - Macroscopic damage (changes in detector properties)
- Approaches to obtain radiation hard sensors
  - Material Engineering
  - Device Engineering
- Summary

# Main motivations for R&D on Radiation Tolerant Detectors: Super - LHC



• LHC upgrade

 $\Rightarrow$  LHC (2007), L = 10<sup>34</sup>cm<sup>-2</sup>s<sup>-1</sup>



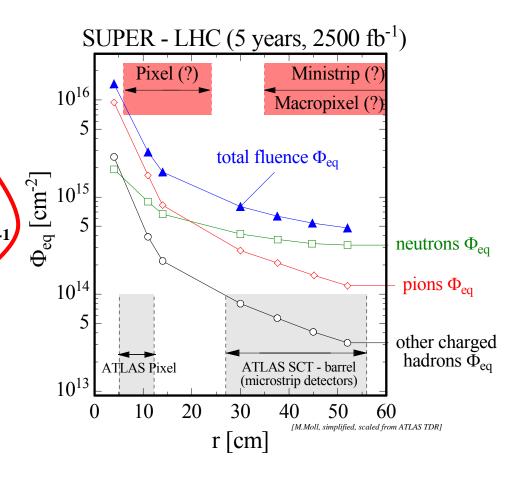
5 years  $\phi(r=4cm) \sim 1.6 \cdot 10^{16} cm^{-2}$ 

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• LHC (Replacement of components)

e.g. - LHCb Velo detectors (~2010)

- ATLAS Pixel B-layer (~2012)



• Linear collider experiments (generic R&D)

Deep understanding of radiation damage will be fruitful for linear collider experiments where high doses of e,  $\gamma$  will play a significant role.

### The CERN RD50 Collaboration

CERN

http://www.cern.ch/rd50

**RD50: Development of Radiation Hard Semiconductor Devices for High Luminosity Colliders** 

- Collaboration formed in November 2001
- Experiment approved as RD50 by CERN in June 2002
- Main objective:

Development of ultra-radiation hard semiconductor detectors for the luminosity upgrade of the LHC to 10<sup>35</sup> cm<sup>-2</sup>s<sup>-1</sup> ("Super-LHC").

Challenges: - Radiation hardness up to 10<sup>16</sup> cm<sup>-2</sup> required

- **Fast signal collection** (Going from 25ns to 10 ns bunch crossing?)
- Low mass (reducing multiple scattering close to interaction point)
- Cost effectiveness (big surfaces have to be covered with detectors!)
- Presently 251 members from 51 institutes

Belarus (Minsk), Belgium (Louvain), Canada (Montreal), Czech Republic (Prague (3x)), Finland (Helsinki, Lappeenranta), Germany (Berlin, Dortmund, Erfurt, Freiburg, Hamburg, Karlsruhe), Israel (Tel Aviv), Italy (Bari, Bologna, Florence, Padova, Perugia, Pisa, Trento, Turin), Lithuania (Vilnius), Norway (Oslo (2x)), Poland (Warsaw(2x)), Romania (Bucharest (2x)), Russia (Moscow), St.Petersburg), Slovenia (Ljubljana), Spain (Barcelona, Valencia), Switzerland (CERN, PSI), Ukraine (Kiev), United Kingdom (Exeter, Glasgow, Lancaster, Liverpool, Sheffield, University of Surrey), USA (Fermilab, Purdue University, Rochester University, SCIPP Santa Cruz, Syracuse University, BNL, University of New Mexico)

## **Radiation Damage in Silicon Sensors**





- Two general types of radiation damage to the detector materials:
  - Bulk (Crystal) damage due to Non Ionizing Energy Loss (NIEL)
    - displacement damage, built up of crystal defects –
    - I. Change of effective doping concentration (higher depletion voltage, under- depletion)
    - II. Increase of leakage current (increase of shot noise, thermal runaway)
    - **III.** Increase of charge carrier trapping (loss of charge)
  - Surface damage due to Ionizing Energy Loss (IEL)
    - accumulation of positive in the oxide (SiO<sub>2</sub>) and the Si/SiO<sub>2</sub> interface affects: interstrip capacitance (noise factor), breakdown behavior, ...
- Impact on detector performance and Charge Collection Efficiency (depending on detector type and geometry and readout electronics!)

Signal/noise ratio is the quantity to watch

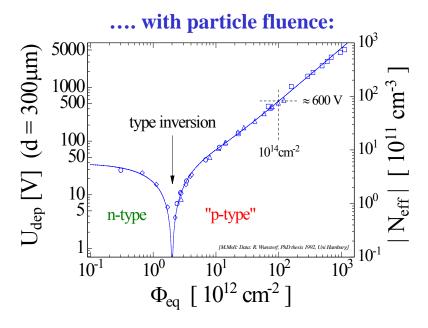
⇒ Sensors can fail from radiation damage!

# RD50 Radiation Damage – I. Effective doping concentration

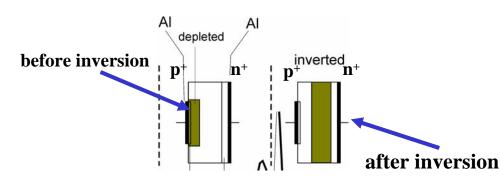


Review (2/5)

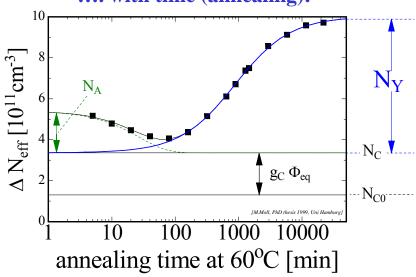
 $\blacksquare \quad \textbf{Change of Depletion Voltage } \mathbf{V}_{\text{dep}} (\mathbf{N}_{\text{eff}})$ 



• "Type inversion": N<sub>eff</sub> changes from positive to negative (Space Charge Sign Inversion)



.... with time (annealing):



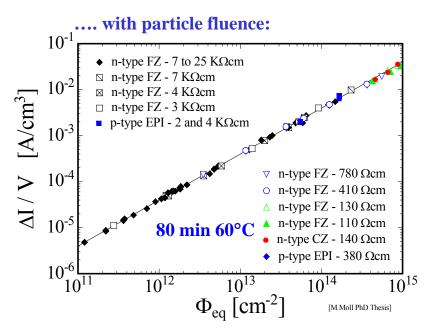
- Short term: "Beneficial annealing"
- Long term: "Reverse annealing"
  - time constant depends on temperature:
    - $\sim 500 \text{ years } (-10^{\circ}\text{C})$
    - $\sim 500 \text{ days} (20^{\circ}\text{C})$
    - $\sim$  21 hours (  $60^{\circ}$ C)
  - Consequence: **Detectors must be cooled** even when the experiment is not running!

# Radiation Damage – II. Leakage Current





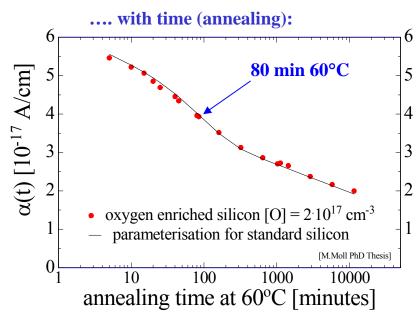
**■** Change of Leakage Current (after hadron irradiation)



• Damage parameter α (slope in figure)

$$\alpha = \frac{\Delta I}{V \cdot \Phi_{eq}}$$
 Leakage current per unit volume and particle fluence

 α is constant over several orders of fluence and independent of impurity concentration in Si
 ⇒ can be used for fluence measurement



- Leakage current decreasing in time (depending on temperature)
- Strong temperature dependence

$$I \propto \exp\left(-\frac{E_g}{2k_BT}\right)$$

#### **Consequence:**

Cool detectors during operation! Example: *I*(-10°C) ~1/16 *I*(20°C)

# **Radiation Damage – III. Trapping**





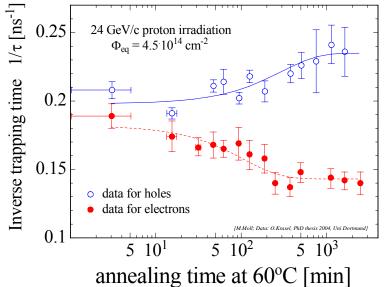
#### Deterioration of Charge Collection Efficiency (CCE) by trapping

**Trapping** is characterized by an effective trapping time  $\tau_{eff}$  for electrons and holes:

$$Q_{e,h}(t) = Q_{0e,h} \exp\left(-\frac{1}{\tau_{eff\ e,h}} \cdot t\right)$$
 where  $\frac{1}{\tau_{eff\ e,h}} \propto N_{defects}$ 

Increase of inverse trapping time  $(1/\tau)$  with fluence ..... and change with time (annealing):

 $1/\tau$  [ns<sup>-1</sup>-24 GeV/c proton irradiation 0.5 0.4 data for electrons o data for holes Inverse trapping time 0.3 0.2 0.1  $4.10^{14}$ particle fluence -  $\Phi_{eq}$  [cm<sup>-2</sup>]



## **Impact on Detector: Decrease of CCE**

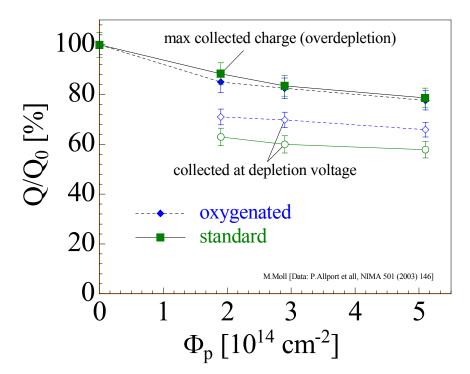




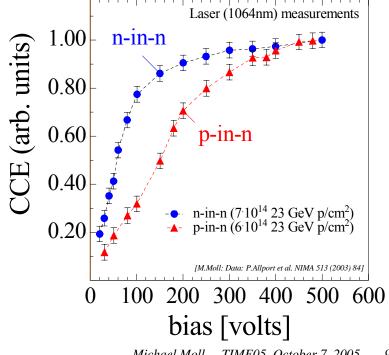
- Loss of signal and increase of noise -
- Two basic mechanisms reduce collectable charge:

  - under-depletion

- ⇒ (depending on detector design and geometry!)
- **Example:** ATLAS microstrip detectors + fast electronics (25ns)
- p-in-n: oxygenated versus standard FZ
  - beta source
  - 20% charge loss after  $5 \times 10^{14}$  p/cm<sup>2</sup> (23 GeV)



- n-in-n versus p-in-n
  - same material, ~ same fluence
  - over-depletion needed



# Approaches to develop radiation harder tracking detectors



## **Scientific strategies:**

- I. Material engineering
- **II.** Device engineering
- III. Change of detector operational conditions

CERN-RD39

"Cryogenic Tracking Detectors"

Talks this Workshop Gianluigi Casse Vincenzo Chiochia • Defect Engineering of Silicon

- Understanding radiation damage
  - Macroscopic effects and Microscopic defects
  - Simulation of defect properties & kinetics
  - Irradiation with different particles & energies
- Oxygen rich Silicon
  - DOFZ, Cz, MCZ, EPI
- Oxygen dimer & hydrogen enriched Si
- Pre-irradiated Si
- Influence of processing technology
- **New Materials** 
  - Silicon Carbide (SiC), Gallium Nitride (GaN)
  - Diamond: CERN RD42 Collaboration
  - Amorphous silicon
- Device Engineering (New Detector Designs)
  - p-type silicon detectors (n-in-p)
  - thin detectors
  - 3D and Semi 3D detectors
  - Stripixels
  - Cost effective detectors
  - Simulation of highly irradiated detectors
  - Monolithic devices

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# **RD50** Sensor Materials: SiC and GaN



Property	Diamond	GaN	4H SiC	Si
E <sub>g</sub> [eV]	5.5	3.39	3.26	1.12
E <sub>breakdown</sub> [V/cm]	$10^7$	$4.10^{6}$	$2.2 \cdot 10^6$	$3.10^{5}$
$\mu_{\rm e}  [{\rm cm}^2/{\rm Vs}]$	1800	1000	800	1450
$\mu_h [cm^2/Vs]$	1200	30	115	450
v <sub>sat</sub> [cm/s]	$2.2 \cdot 10^7$	-	$2 \cdot 10^7$	$0.8 \cdot 10^{7}$
Z	6	31/7	14/6	14
$\mathcal{E}_{\mathrm{r}}$	5.7	9.6	9.7	11.9
e-h energy [eV]	13	8.9	7.6-8.4	3.6
Density [g/cm3]	3.515	6.15	3.22	2.33
Displacem. [eV]	43	≥15	25	13-20

- Wide bandgap (3.3eV)
- ⇒ lower leakage current than silicon
- Signal:
  Diamond 36 e/μm
  SiC 51 e/μm
  Si 89 e/μm
- ⇒ more charge than diamond

R&D on diamond detectors: RD42 – Collaboration http://cern.ch/rd42/

- Higher displacement threshold than silicon
- ⇒ radiation harder than silicon (?)

**Recent review:** P.J.Sellin and J.Vaitkus on behalf of RD50 "New materials for radiation hard semiconductor detectors", submitted to NIMA

# SiC: CCE after irradiation



#### **CCE** before irradiation

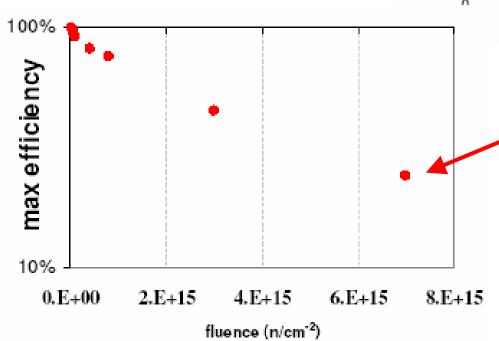
- before irrau...

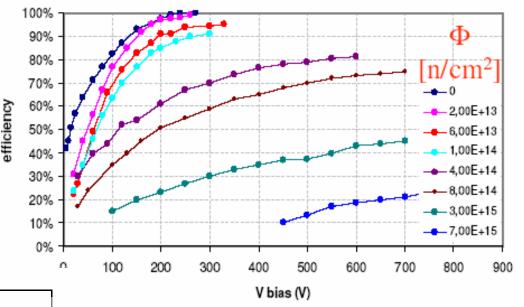
  100 % with α particles and iv...

  tested on various samples 20-40μm

#### **CCE** after irradiation

- with  $\alpha$  particles
- neutron irradiated samples
- material produced by CREE
- 25 µm thick layer





[S.Sciortino et al., presented on the RESMDD 04 conference, in press with NIMA ]

20% CCE (α) after 7x10<sup>15</sup> n/cm<sup>2</sup>!

35% CCE(β) (CCD ~6μm; ~ 300 e) after  $1.4 \times 10^{16} \text{ p/cm}^2$ 

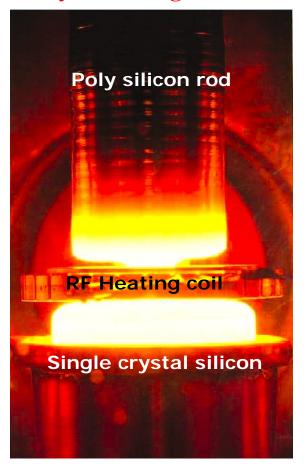
much less than in silicon (see later)

# **Material: Float Zone Silicon (FZ)**



## **■ Float Zone process**

• Using a single Si crystal seed, melt the vertically oriented rod onto the seed using RF power and "pull" the monocrystalline ingot



## **■ Mono-crystalline Ingot**



## **■ Wafer production**

• Slicing, lapping, etching, polishing



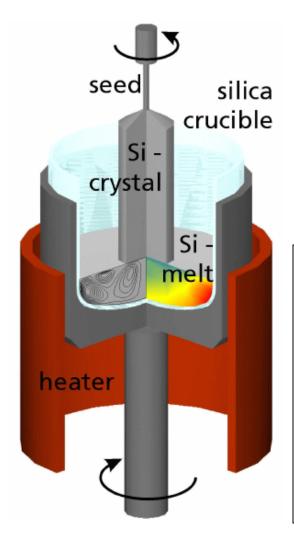
## **■** Oxygen enrichment (DOFZ)

• Oxidation of wafer at high temperatures

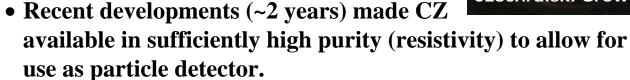
# RD50 Czochralski silicon (Cz) & Epitaxial silicon (EPI)



#### Czochralski silicon



- Pull Si-crystal from a Si-melt contained in a silica crucible while rotating.
- Silica crucible is dissolving oxygen into the melt ⇒ high concentration of O in CZ
- Material used by IC industry (cheap)





### Epitaxial silicon

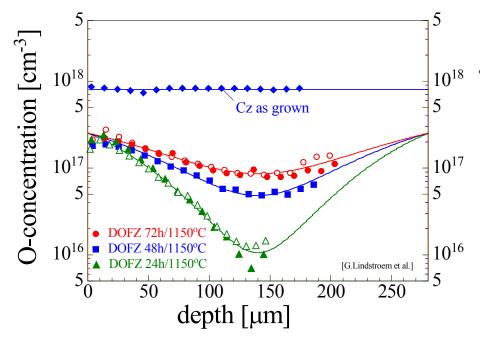
- Chemical-Vapor Deposition (CVD) of Silicon
- CZ silicon substrate used ⇒ in-diffusion of oxygen
- growth rate about 1μm/min
- excellent homogeneity of resistivity
- up to 150 μm thick layers produced
- price depending on thickness of epi-layer but not extending ~ 3 x price of FZ wafer

## Oxygen concentration in FZ, CZ and EPI



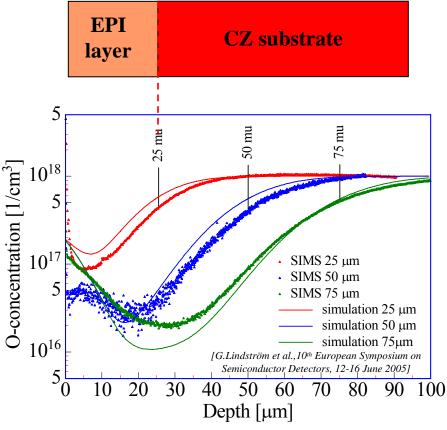
#### **■ Cz and DOFZ silicon**

- CZ: high O<sub>i</sub> (oxygen) and O<sub>2i</sub> (oxygen dimer) concentration (homogeneous)
- CZ: formation of Thermal Donors possible!



- DOFZ: inhomogeneous oxygen distribution
- DOFZ: oxygen content increasing with time at high temperature

## **Epitaxial silicon**



- EPI: O<sub>i</sub> and O<sub>2i</sub> (?) diffusion from substrate into epi-layer during production
- EPI: in-homogeneous oxygen distribution

# Standard FZ, DOFZ, Cz and MCz Silicon



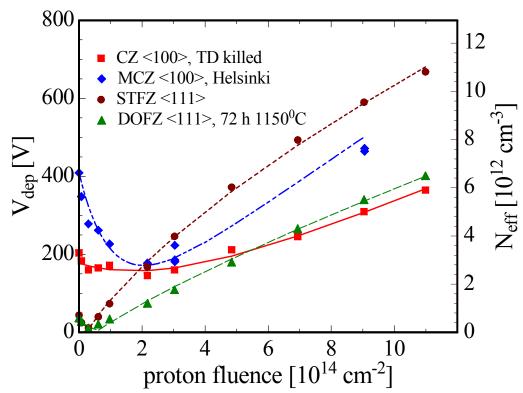
## 24 GeV/c proton irradiation

#### Standard FZ silicon

- type inversion at  $\sim 2 \times 10^{13} \text{ p/cm}^2$
- strong  $N_{eff}$  increase at high fluence

### Oxygenated FZ (DOFZ)

- type inversion at  $\sim 2 \times 10^{13} \text{ p/cm}^2$
- reduced  $N_{\text{eff}}$  increase at high fluence



#### CZ silicon and MCZ silicon

- <u>no type inversion</u> in the overall fluence range (verified by TCT measurements) (verified for CZ silicon by TCT measurements, preliminary result for MCZ silicon)
  - ⇒ donor generation overcompensates acceptor generation in high fluence range

#### • Common to all materials (after hadron irradiation):

- reverse current increase
- increase of trapping (electrons and holes) within ~ 20%

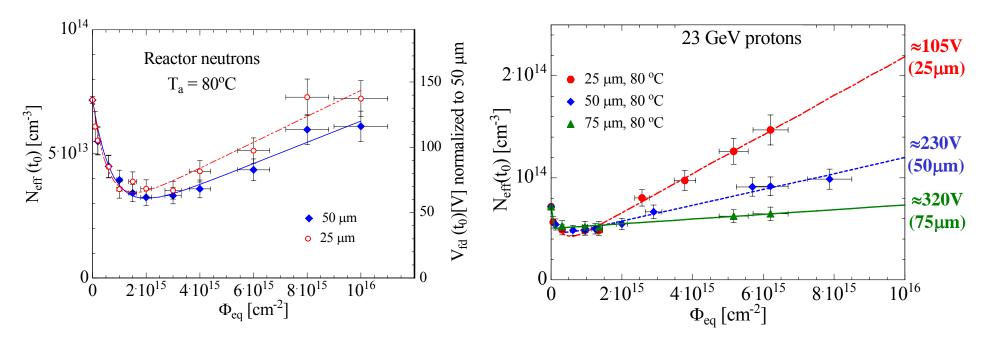
## **EPI Devices – Irradiation experiments**



Epitaxial silicon grown by ITME

G.Lindström et al., 10<sup>th</sup> European Symposium on Semiconductor Detectors, 12-16 June 2005

- Layer thickness: 25, 50, 75  $\mu$ m; resistivity: ~ 50  $\Omega$ cm
- Oxygen:  $[O] \approx 9 \times 10^{16} \text{cm}^{-3}$ ; Oxygen dimers (detected via  $IO_2$ -defect formation)

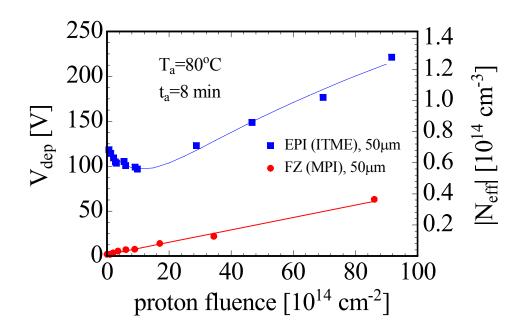


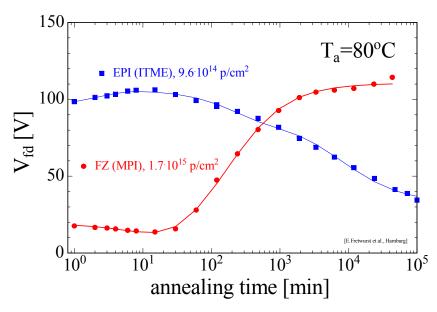
- No type inversion in the full range up to  $\sim 10^{16}$  p/cm<sup>2</sup> and  $\sim 10^{16}$  n/cm<sup>2</sup> (type inversion only observed during long term annealing)
- Proposed explanation: introduction of shallow donors bigger than generation of deep acceptors

# **Epitaxial silicon - Annealing**



- 50 μm thick silicon detectors:
  - Epitaxial silicon (50Ωcm on CZ substrate, ITME & CiS)
  - Thin FZ silicon (4KΩcm, MPI Munich, wafer bonding technique)





[E.Fretwurst et al., RESMDD - October 2004]

- Thin FZ silicon: Type inverted, increase of depletion voltage with time
- Epitaxial silicon: No type inversion, decrease of depletion voltage with time

 $\Rightarrow$  No need for low temperature during maintenance of SLHC detectors!



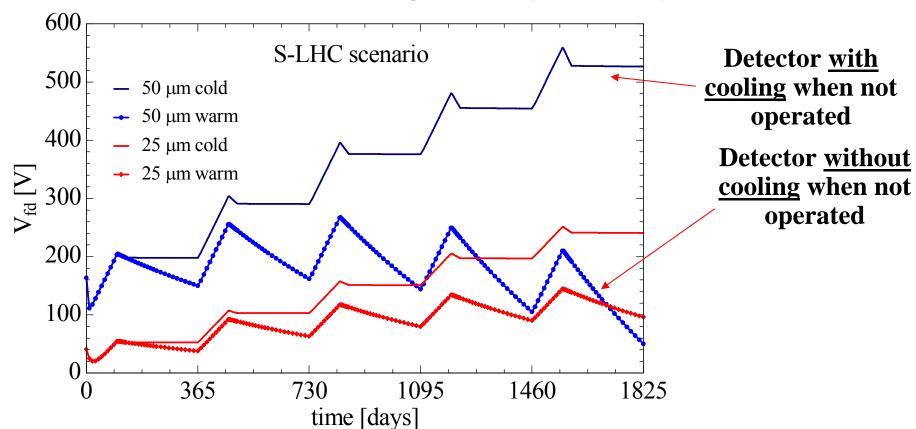
## **Damage Projection – SLHC**



- 50 μm EPI silicon: a solution for pixels ?-

G.Lindström et al., 10th European Symposium on Semiconductor Detectors, 12-16 June 2005 (Damage projection: M.Moll)

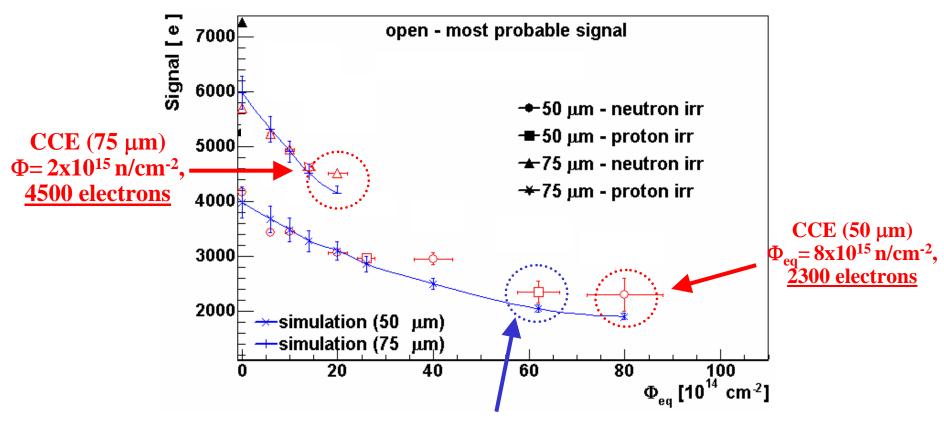
- **Radiation level (4cm):**  $\Phi_{eq}(year) = 3.5 \times 10^{15} \text{ cm}^{-2}$
- SLHC-scenario: 1 year = 100 days beam (-7°C)
  30 days maintenance (20°C)
  235 days no beam (-7°C or 20°C)



# Signal from irradiated EPI



- Epitaxial silicon: CCE measured with beta particles (90Sr)
  - 25ns shaping time
  - proton and neutron irradiations of 50 μm and 75 μm epi layers

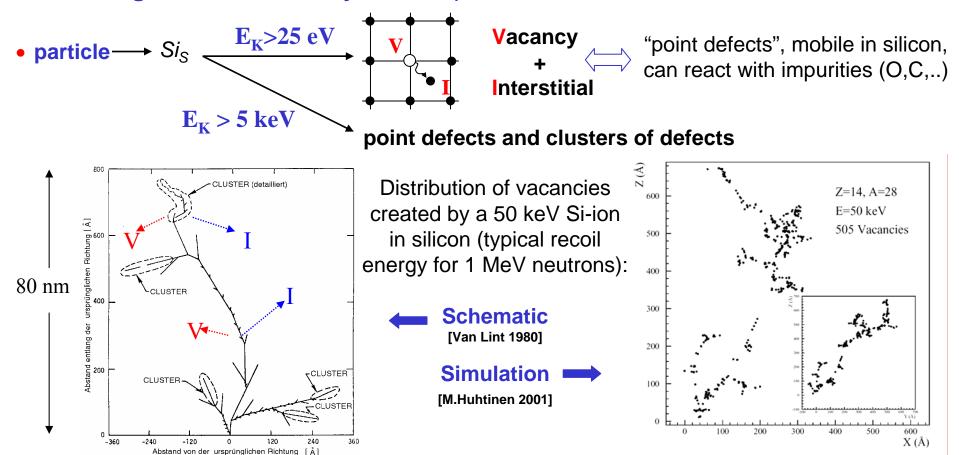


CCE (50  $\mu$ m):  $\Phi$ = 1x10<sup>16</sup>cm<sup>-2</sup> (24GeV/c protons) 2400 electrons

# **Microscopic defects**



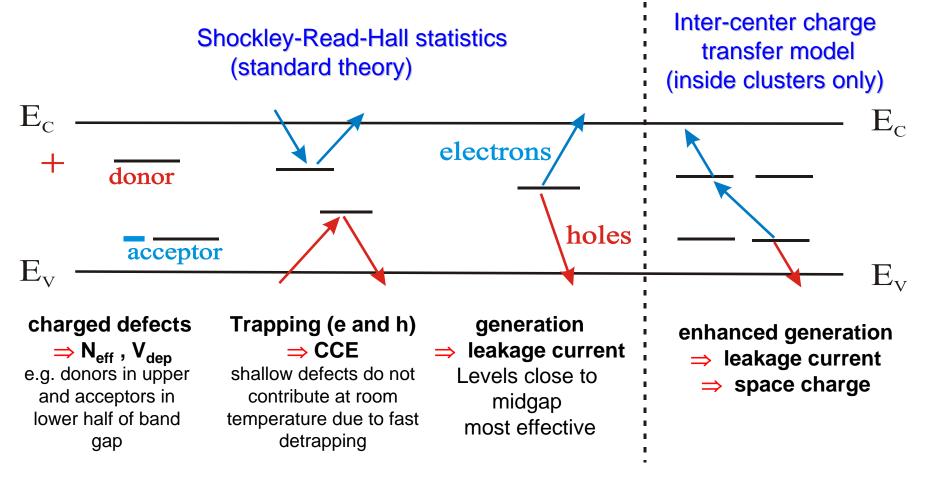
Damage to the silicon crystal: Displacement of lattice atoms



- Defects can be electrically active (levels in the band gap)
  - capture and release electrons and holes from conduction and valence band
  - ⇒ can be charged can be generation/recombination centers can be trapping centers

# RD50 Impact of Defects on Detector properties





Impact on detector properties can be calculated if all defect parameters are known:

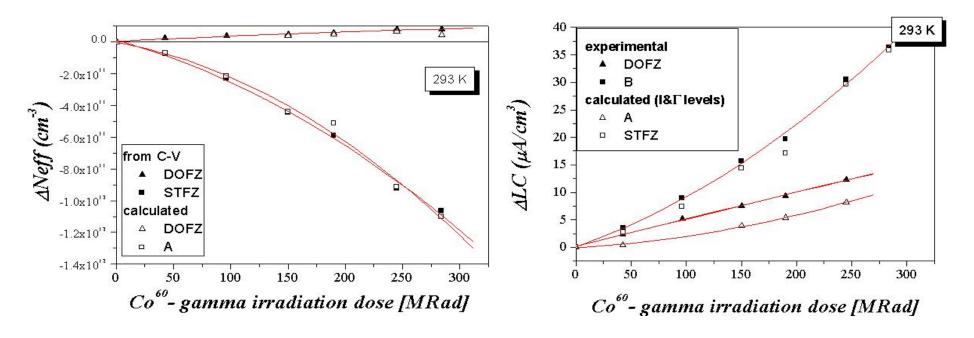
 $\sigma_{n,p}$ : cross sections  $\Delta E$ : ionization energy  $N_t$ : concentration

# 

(CERN)

- Co<sup>60</sup> γ-irradiated silicon detectors -

- Comparison for effective doping concentration (left) and leakage current (right) for two different materials
  - as predicted by the microscopic measurements (open symbols)
  - as deduced from CV/IV characteristics (filled symbols)



[I.Pintilie et al., Applied Physics Letters, 82, 2169, March 2003]

# Characterization of microscopic defects



- γ and proton irradiated silicon detectors -

- 2003: Major breakthrough on γ-irradiated samples
  - For the first time macroscopic changes of the <u>depletion voltage and leakage current</u> can be explained by electrical properties of measured defects! [APL, 82, 2169, March 2003]
- since 2004: Big step in understanding the improved radiation tolerance of oxygen enriched and epitaxial silicon after proton irradiation

# Levels responsible for depletion voltage changes after proton irradiation:

Almost independent of oxygen content:

- Donor removal
- "Cluster damage" ⇒ negative charge

#### **Influenced by initial oxygen content:**

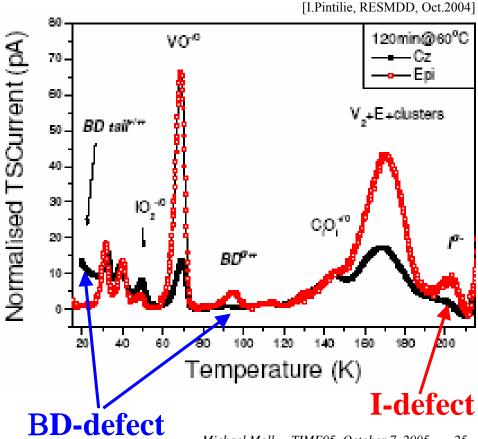
• I-defect: deep acceptor level at  $E_C$ -0.54eV (good candidate for the  $V_2O$  defect)

**⇒** negative charge

#### Influenced by <u>initial oxygen dimer</u> content (?):

• **BD-defect:** bistable shallow thermal  $\underline{\text{donor}}$  (formed via oxygen dimers  $O_{2i}$ )

**⇒** positive charge



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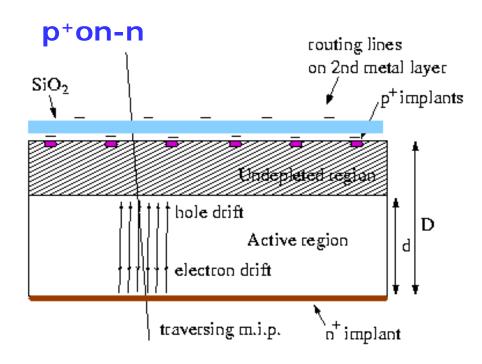
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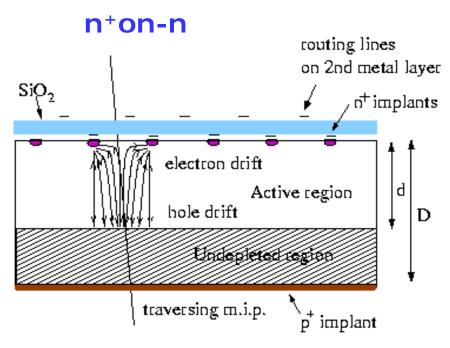
# **Device engineering**

## p-in-n versus n-in-n detectors



## n-type silicon after type inversion:





#### p-on-n silicon, under-depleted:

- Charge spread degraded resolution
- Charge loss reduced CCE

#### n-on-n silicon, under-depleted:

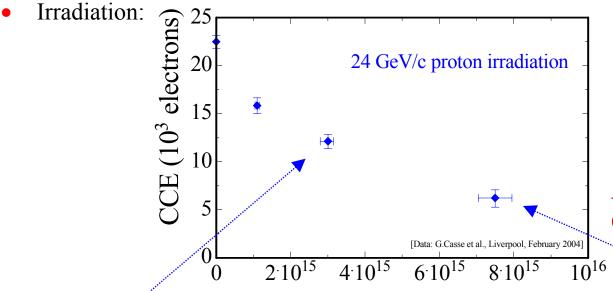
- Limited loss in CCE
- •Less degradation with under-depletion
- Collect electrons (fast)

# n-in-p microstrip detectors



n-in-p: - no type inversion, high electric field stays on structured side- collection of electrons

- Miniature n-in-p microstrip detectors (280μm)
- Detectors read-out with LHC speed (40MHz) chip (SCT128A)
- Material: standard p-type and oxygenated (DOFZ) p-type



G. Casse et al., NIMA535(2004) 362

At the highest fluence Q~6500e at V<sub>bias</sub>=900V

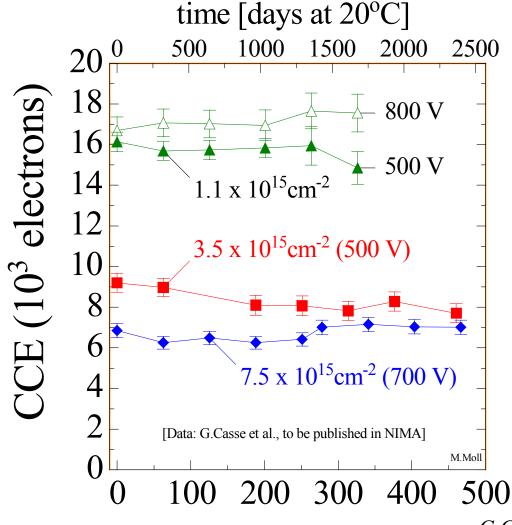
CCE  $\sim 60\%$  after 3  $10^{15}$  p cm<sup>-2</sup> at 900V( standard p-type)

fluence [cm<sup>-2</sup>]

CCE ~ 30% after 7.5  $10^{15}$  p cm<sup>-2</sup> 900V (oxygenated p-type)

# Annealing of p-type sensors





- p-type strip detector (280 $\mu$ m) irradiated with 23 GeV p (7.5 × 10<sup>15</sup> p/cm<sup>2</sup>)
- expected from previous CV measurement of  $V_{dep}$ :
  - before reverse annealing:

$$V_{dep} \sim 2800V$$

- after reverse annealing

$$V_{dep} > 12000V$$

• no reverse annealing visible in the CCE measurement!

time at 80°C[min]

G. Casse et al., 10<sup>th</sup> European Symposium on Semiconductor Detectors, 12-16 June 2005

## **Device Engineering: 3D detectors**



#### • Electrodes:

(Introduced by S.I. Parker et al., NIMA 395 (1997) 328)

- narrow columns along detector thickness-"3D"
- diameter: 10μm distance: 50 100μm

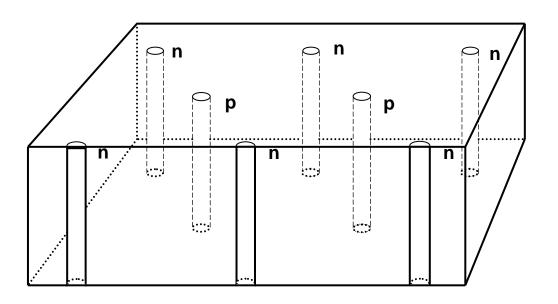
#### • Lateral depletion:

- lower depletion voltage needed
- thicker detectors possible
- fast signal

#### Hole processing :

- Dry etching, Laser drilling, Photo Electro Chemical
- Present aspect ratio (RD50) 30:1

Production of 3D sensor matched to ATLAS Pixel readout chip under way (S.Parker, Pixel 2005)



## **Device Engineering: 3D detectors**



#### • Electrodes:

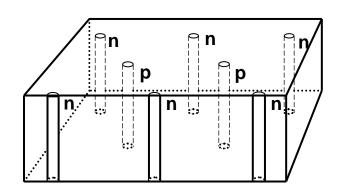
(Introduced by S.I. Parker et al., NIMA 395 (1997) 328)

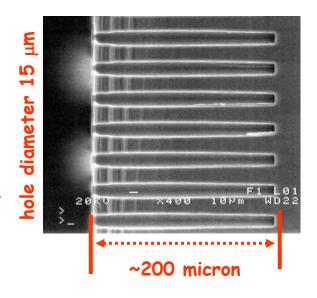
- narrow columns along detector thickness-"3D"
- diameter: **10μm** distance: **50 100μm**
- Lateral depletion:
  - lower depletion voltage needed
  - thicker detectors possible
  - fast signal
- Hole processing :
  - Dry etching, Laser drilling, Photo Electro Chemical
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#### 3D detector developments within RD50:

- 1) Glasgow University pn junction & Schottky contacts Irradiation tests up to  $5 \times 10^{14}$  p/cm<sup>2</sup> and  $5 \times 10^{14}$  π/cm<sup>2</sup>:  $V_{fd} = 19 \text{V}$  (inverted); CCE drop by 25% ( $\alpha$ -particles)
- 2) IRST-Trento and CNM Barcelona (since 2003)

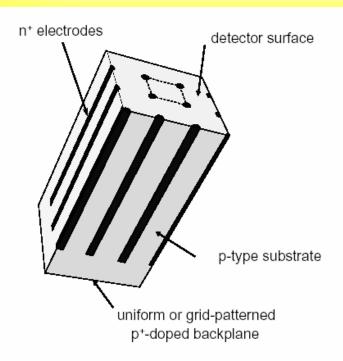
CNM: Hole etching (DRIE); IRST: all further processing diffused contacts or doped polysilicon deposition





#### **RD50** 3D Detectors: New Architecture





### **Fabrication planned for end 2005**

INFN/Trento funded project: collaboration between IRST, Trento and CNM Barcelona

#### **Simulation**

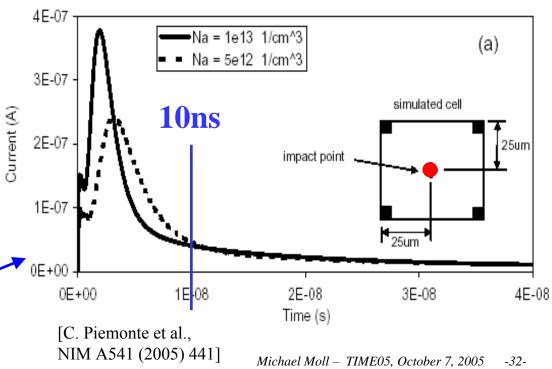
- CCE within < 10 ns
- worst case shown (hit in middle of cell)

#### Simplified 3D architecture

- n<sup>+</sup> columns in p-type substrate, p<sup>+</sup> backplane
- operation similar to standard 3D detector

#### **Simplified process**

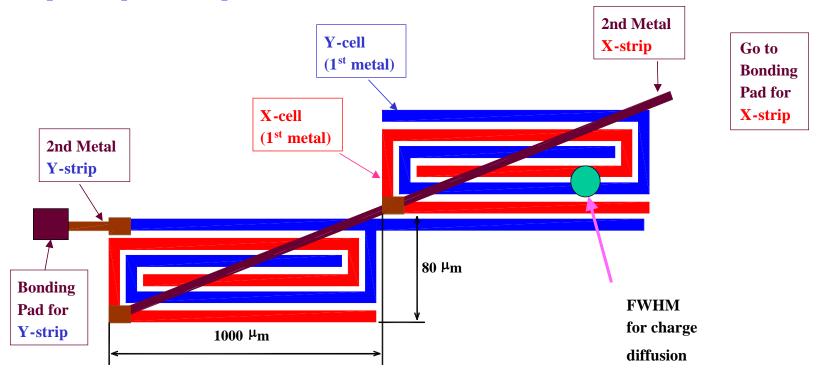
- hole etching and doping only done once
- no wafer bonding technology needed



## **Example for new structures - Stripixel**



- New structures: There is a multitude of concepts for new (planar and mixed planar & 3D) detector structures aiming for improved radiation tolerance or less costly detectors (see e.g. Z.Li 6<sup>th</sup> RD50 workshop)
- **Example: Stripixel concept:**



Z. Li, D. Lissauer, D. Lynn, P. O'Connor, V. Radeka

# **Summary**



- At fluences up to 10<sup>15</sup>cm<sup>-2</sup> (Outer layers of a SLHC detector) the change of depletion voltage and the large area to be covered by detectors is the major problem.
  - CZ silicon detectors could be a cost-effective radiation hard solution (no type inversion, use p-in-n technology)
  - p-type silicon microstrip detectors show very encouraging results:  $CCE \approx 6500 \text{ e}$ ;  $\Phi_{eq} = 4 \times 10^{15} \text{ cm}^{-2}$ ,  $300 \mu \text{m}$ , collection of electrons, no reverse annealing observed in CCE measurement!
- At the fluence of 10<sup>16</sup>cm<sup>-2</sup> (Innermost layer of a SLHC detector) the active thickness of any silicon material is significantly reduced due to trapping. The promising new options are:

Thin/EPI detectors: drawback: radiation hard electronics for low signals needed e.g. 2300e at Φ<sub>eq</sub> 8x10<sup>15</sup>cm<sup>-2</sup>, 50μm EPI, .... thicker layers will be tested in 2005/2006

3D detectors: drawback: technology has to be optimized ..... steady progress within RD50

New Materials like SiC and GaN (not shown) have been characterized.
 CCE tests show that these materials are not radiation harder than silicon

Info: http://cern.ch/rd50; 7th RD50 Workshop at CERN: 14-16 November