

PIXEL 2005 international workshop, September 5-8, Bonn, Germany

Radiation Tolerant Sensors for Pixel Detectors

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CERN - Geneva - Switzerland

on behalf of the

- CERN-RD50 project –

<http://www.cern.ch/rd50>



- **Motivation to develop radiation harder detectors: Super-LHC**
- **Introduction to the RD50 collaboration**
- **Radiation Damage in Silicon Detectors (A review in 5 slides)**
 - **Macroscopic damage (changes in detector properties)**
- **Approaches to obtain radiation hard sensors**
 - **Material Engineering**
 - **Device Engineering**
- **Summary**



- LHC upgrade**

⇒ LHC (2007), $L = 10^{34} \text{cm}^{-2}\text{s}^{-1}$

10 years
500 fb⁻¹

$$\phi(r=4\text{cm}) \sim 3 \cdot 10^{15} \text{cm}^{-2}$$

× 5

⇒ Super-LHC (2015 ?), $L = 10^{35} \text{cm}^{-2}\text{s}^{-1}$

5 years
2500 fb⁻¹

$$\phi(r=4\text{cm}) \sim 1.6 \cdot 10^{16} \text{cm}^{-2}$$

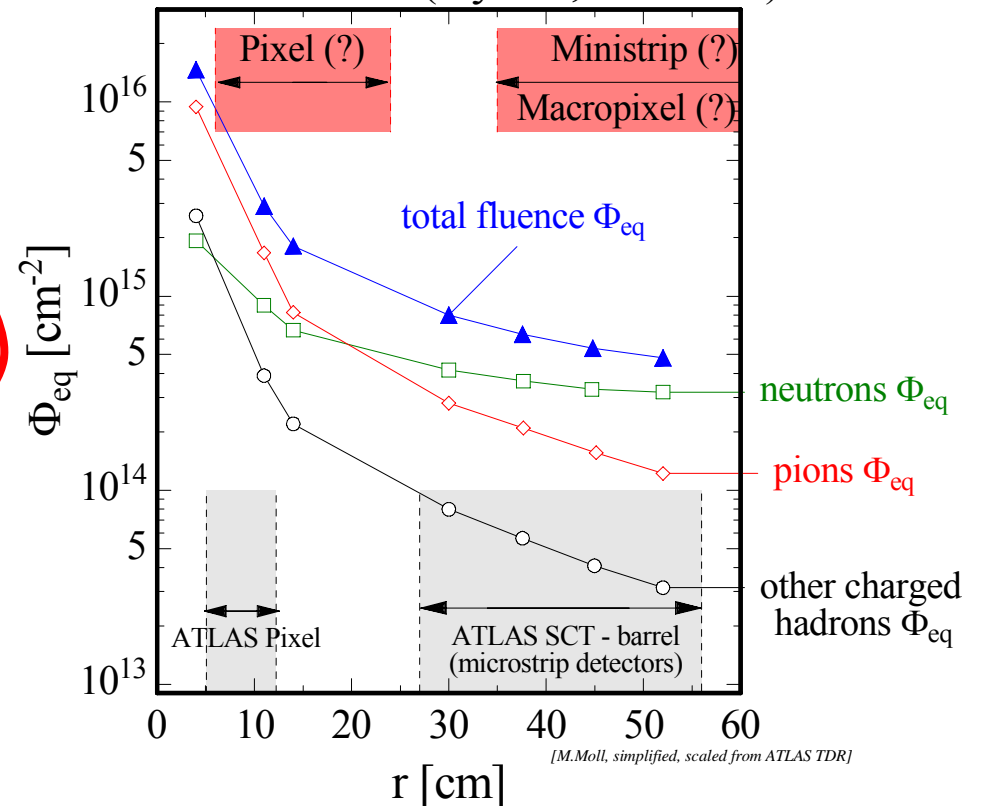
- LHC (Replacement of components)**

- e.g. - LHCb Velo detectors (~2010)
- ATLAS Pixel B-layer (~2012)

- Linear collider experiments (generic R&D)**

Deep understanding of radiation damage will be fruitful for linear collider experiments where high doses of e, γ will play a significant role.

SUPER - LHC (5 years, 2500 fb⁻¹)





RD50: Development of Radiation Hard Semiconductor Devices for High Luminosity Colliders

- Collaboration formed in November 2001
- Experiment approved as RD50 by CERN in June 2002
- Main objective:

Development of ultra-radiation hard semiconductor detectors for the luminosity upgrade of the LHC to $10^{35} \text{ cm}^{-2}\text{s}^{-1}$ (“Super-LHC”).

Challenges:

- Radiation hardness up to 10^{16} cm^{-2} required
- Fast signal collection (Going from 25ns to 10 ns bunch crossing ?)
- Low mass (reducing multiple scattering close to interaction point)
- Cost effectiveness (big surfaces have to be covered with detectors!)

- Presently 251 members from 51 institutes

Belarus (Minsk), **Belgium** (Louvain), **Canada** (Montreal), **Czech Republic** (Prague (3x)), **Finland** (Helsinki, Lappeenranta), **Germany** (Berlin, Dortmund, Erfurt, Freiburg, Hamburg, Karlsruhe), **Israel** (Tel Aviv), **Italy** (Bari, Bologna, Florence, Padova, Perugia, Pisa, Trento, Turin), **Lithuania** (Vilnius), **Norway** (Oslo (2x)), **Poland** (Warsaw(2x)), **Romania** (Bucharest (2x)), **Russia** (Moscow), St.Petersburg), **Slovenia** (Ljubljana), **Spain** (Barcelona, Valencia), **Switzerland** (CERN, PSI), **Ukraine** (Kiev), **United Kingdom** (Exeter, Glasgow, Lancaster, Liverpool, Sheffield, University of Surrey), **USA** (Fermilab, Purdue University, Rochester University, SCIPP Santa Cruz, Syracuse University, BNL, University of New Mexico)



- **Two general types of radiation damage to the detector materials:**
 - **Bulk (Crystal) damage due to Non Ionizing Energy Loss (NIEL)**
 - displacement damage, built up of crystal defects –
 - I. Change of **effective doping concentration** (higher depletion voltage, under- depletion)
 - II. Increase of **leakage current** (increase of shot noise, thermal runaway)
 - III. Increase of **charge carrier trapping** (loss of charge)
 - **Surface damage due to Ionizing Energy Loss (IEL)**
 - accumulation of positive in the oxide (SiO_2) and the Si/SiO_2 interface –
affects: interstrip capacitance (noise factor), breakdown behavior, ...
- **Impact on detector performance and Charge Collection Efficiency**
(depending on detector type and geometry and readout electronics!)

Signal/noise ratio is the quantity to watch

⇒ Sensors can fail from radiation damage !

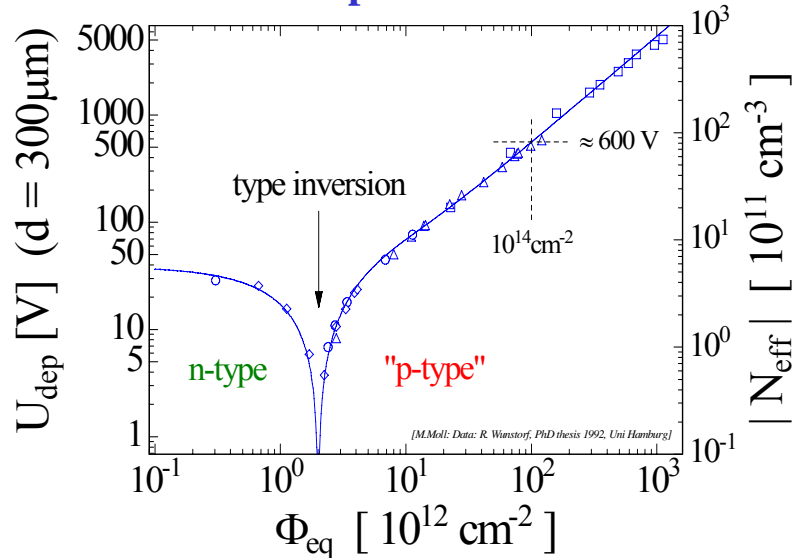
RD50 Radiation Damage – I. Effective doping concentration



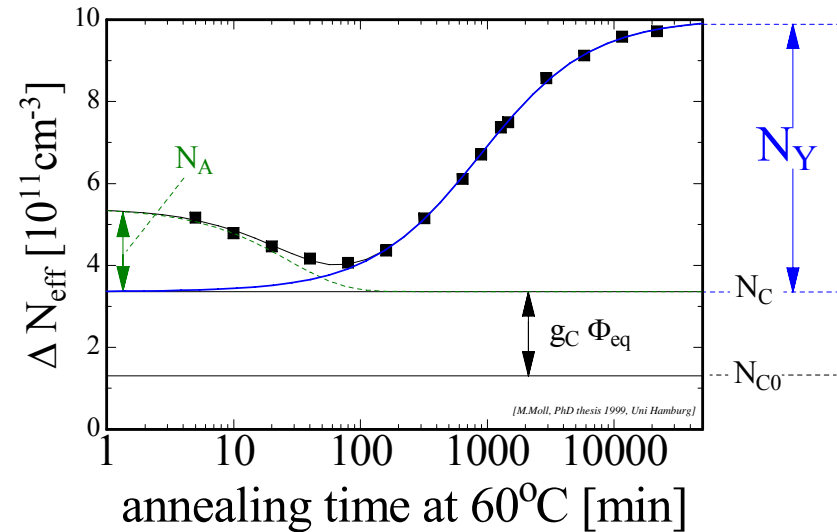
Review
(2/5)

Change of Depletion Voltage V_{dep} (N_{eff})

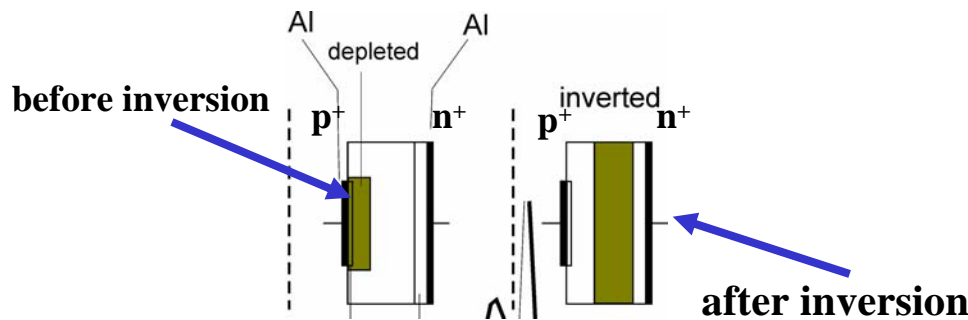
.... with particle fluence:



.... with time (annealing):



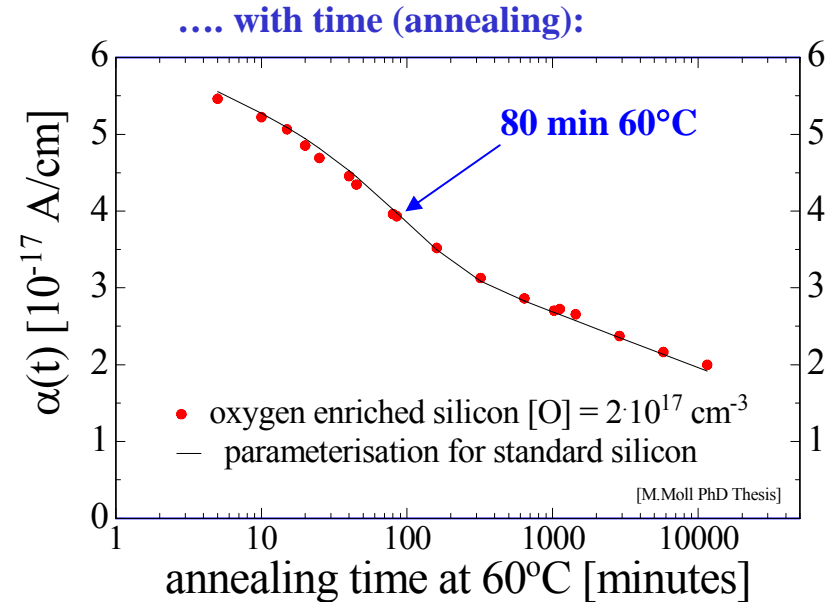
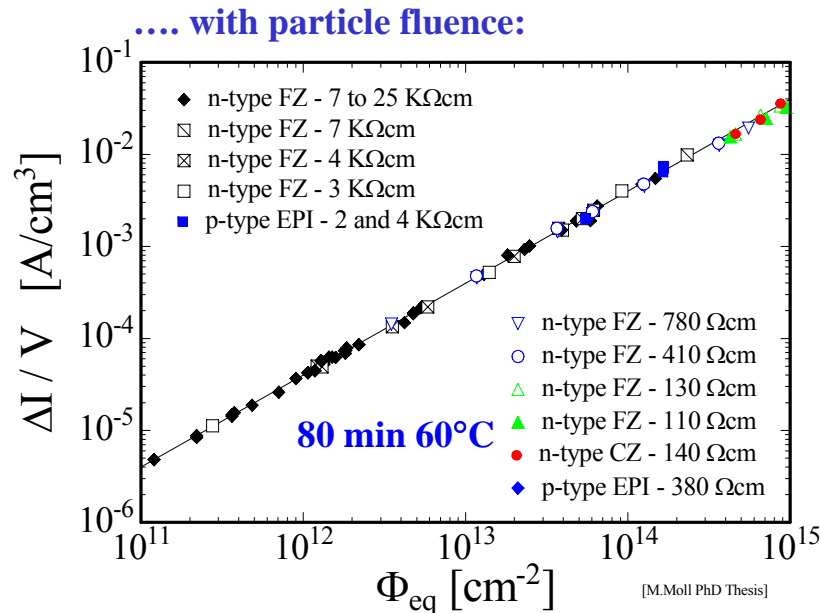
- “**Type inversion**”: N_{eff} changes from positive to negative (Space Charge Sign Inversion)



- Short term: “**Beneficial annealing**”
- Long term: “**Reverse annealing**”
- time constant depends on temperature:
 - ~ 500 years (-10°C)
 - ~ 500 days (20°C)
 - ~ 21 hours (60°C)
- Consequence: **Detectors must be cooled even when the experiment is not running!**



Change of Leakage Current (after hadron irradiation)



- Damage parameter α (slope in figure)

$$\alpha = \frac{\Delta I}{V \cdot \Phi_{eq}}$$

Leakage current
per unit volume
and particle fluence

- α is constant over several orders of fluence and independent of impurity concentration in Si
⇒ can be used for fluence measurement

- Leakage current decreasing in time (depending on temperature)
- Strong temperature dependence

$$I \propto \exp\left(-\frac{E_g}{2k_B T}\right)$$

Consequence:

Cool detectors during operation!
Example: $I(-10^\circ\text{C}) \sim 1/16 I(20^\circ\text{C})$

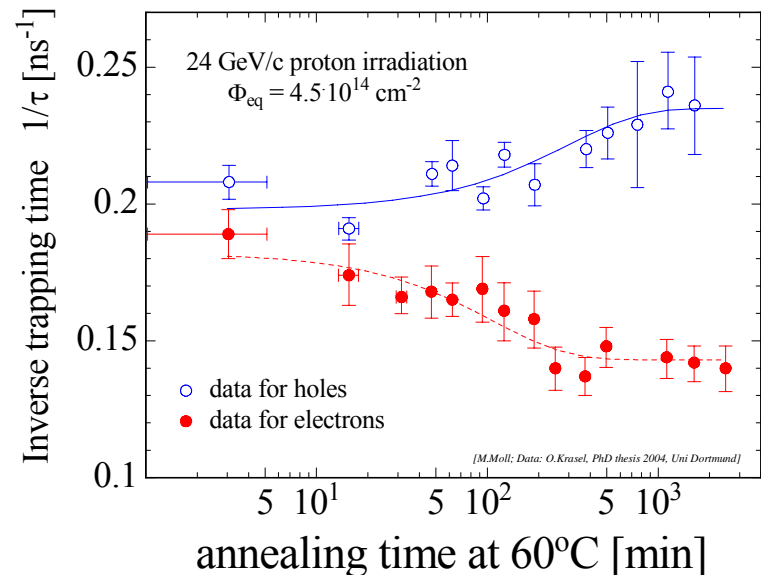
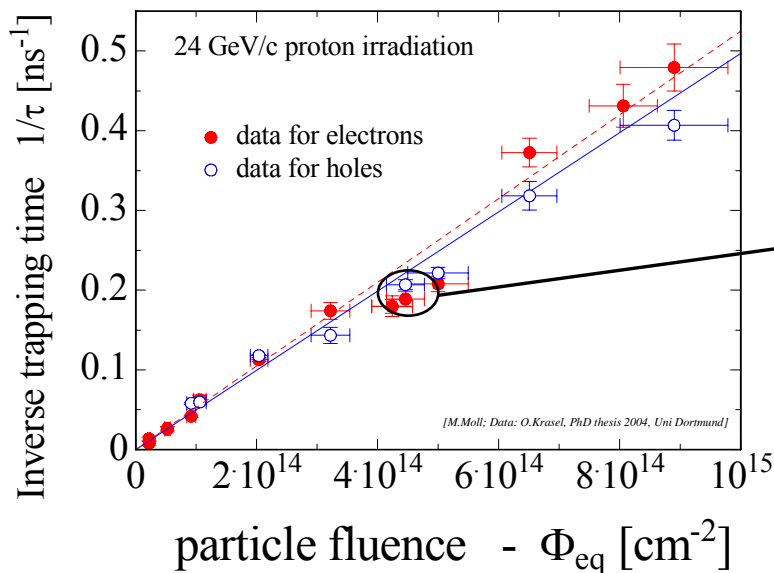


■ Deterioration of Charge Collection Efficiency (CCE) by trapping

Trapping is characterized by an effective trapping time τ_{eff} for electrons and holes:

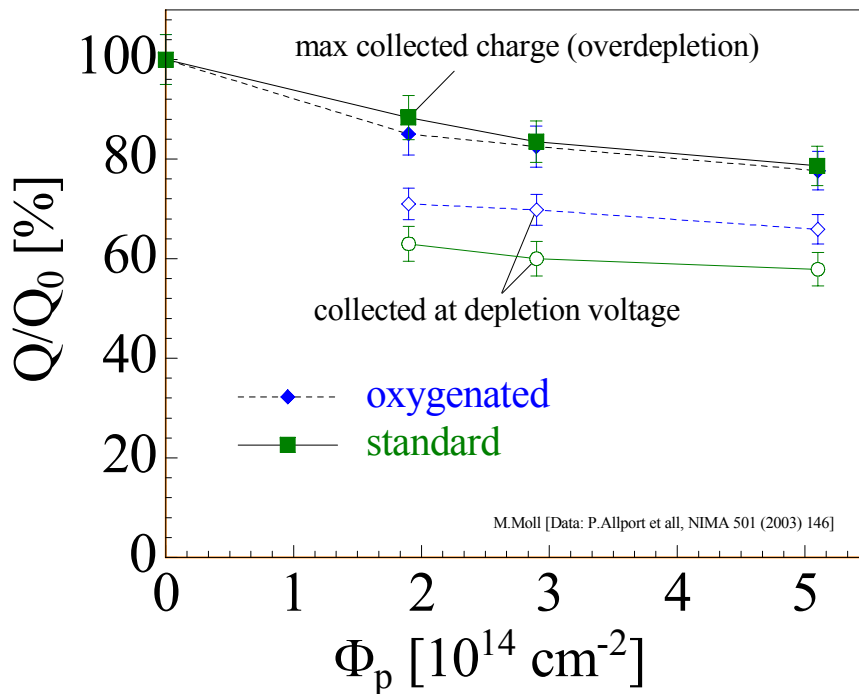
$$Q_{e,h}(t) = Q_{0e,h} \exp\left(-\frac{1}{\tau_{\text{eff } e,h}} \cdot t\right) \quad \text{where} \quad \frac{1}{\tau_{\text{eff } e,h}} \propto N_{\text{defects}}$$

Increase of inverse trapping time ($1/\tau$) with fluence and change with time (annealing):

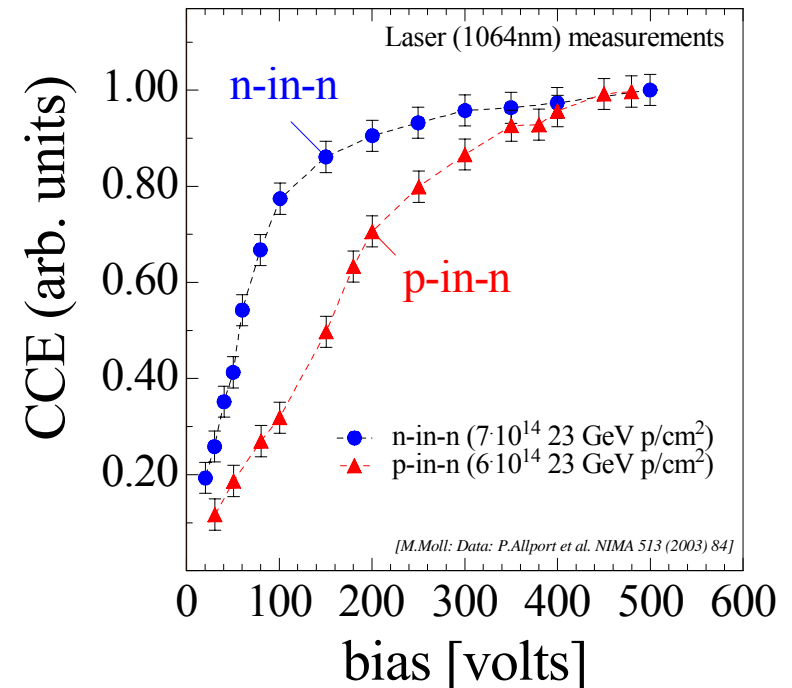




- **Two basic mechanisms reduce collectable charge:**
 - trapping of electrons and holes \Rightarrow (depending on drift and shaping time !)
 - under-depletion \Rightarrow (depending on detector design and geometry !)
- **Example: ATLAS microstrip detectors + fast electronics (25ns)**
- **p-in-n : oxygenated versus standard FZ**
 - beta source
 - 20% charge loss after 5×10^{14} p/cm² (23 GeV)



- **n-in-n versus p-in-n**
 - same material, ~ same fluence
 - over-depletion needed





Scientific strategies:

- I. Material engineering**
- II. Device engineering**
- III. Change of detector operational conditions**

CERN-RD39
“Cryogenic Tracking Detectors”

Talks this Workshop
H.Kagan, R.Stone
D.Moraes
S.Parker
M.Swartz

- **Defect Engineering of Silicon**

- Understanding radiation damage
 - Macroscopic effects and Microscopic defects
 - Simulation of defect properties & kinetics
 - Irradiation with different particles & energies
- Oxygen rich Silicon
 - DOFZ, Cz, MCZ, EPI
- Oxygen dimer & hydrogen enriched Si
- Pre-irradiated Si
- Influence of processing technology

- **New Materials**

- Silicon Carbide (SiC), Gallium Nitride (GaN)
- Diamond: CERN RD42 Collaboration
- Amorphous silicon

- **Device Engineering (New Detector Designs)**

- p-type silicon detectors (n-in-p)
- thin detectors
- 3D and Semi 3D detectors
- Stripixels
- Cost effective detectors
- Simulation of highly irradiated detectors
- Monolithic devices



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 - Device Engineering
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Property	Diamond	GaN	4H SiC	Si
E_g [eV]	5.5	3.39	3.26	1.12
$E_{\text{breakdown}}$ [V/cm]	10^7	$4 \cdot 10^6$	$2.2 \cdot 10^6$	$3 \cdot 10^5$
μ_e [cm^2/Vs]	1800	1000	800	1450
μ_h [cm^2/Vs]	1200	30	115	450
v_{sat} [cm/s]	$2.2 \cdot 10^7$	-	$2 \cdot 10^7$	$0.8 \cdot 10^7$
Z	6	31/7	14/6	14
ϵ_r	5.7	9.6	9.7	11.9
e-h energy [eV]	13	8.9	7.6-8.4	3.6
Density [g/cm ³]	3.515	6.15	3.22	2.33
Displacem. [eV]	43	≥ 15	25	13-20

- Wide bandgap (3.3eV)
- ⇒ lower leakage current than silicon

- Signal:
- Diamond 36 e/ μm
- SiC 51 e/ μm
- Si 89 e/ μm

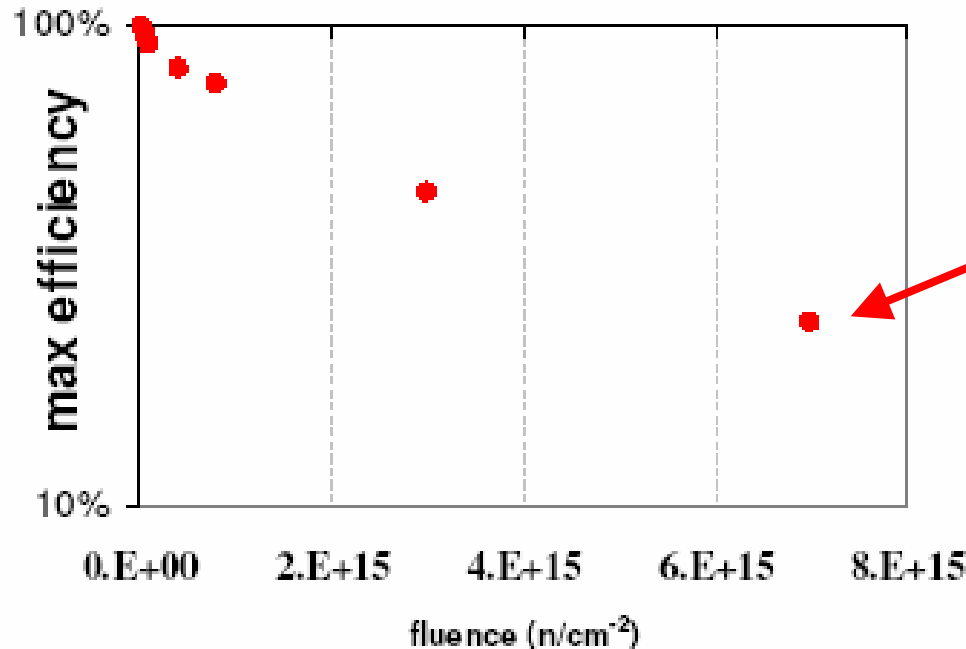
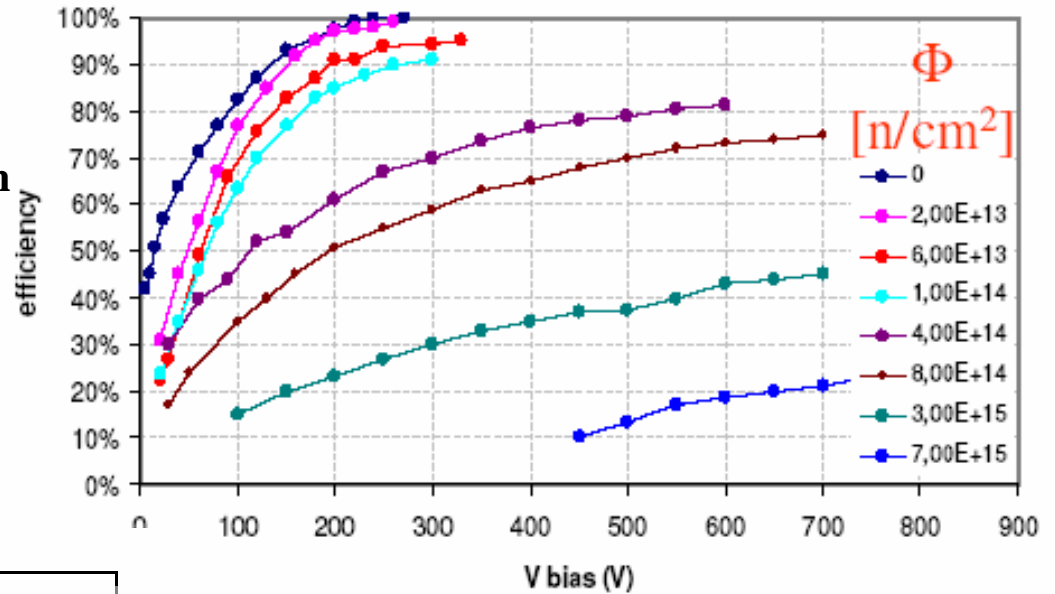
- ⇒ more charge than diamond

- Higher displacement threshold than silicon
- ⇒ radiation harder than silicon (?)

R&D on diamond detectors:
RD42 – Collaboration
<http://cern.ch/rd42/>

Recent review: P.J.Sellin and J.Vaitkus on behalf of RD50 “New materials for radiation hard semiconductor detectors”, submitted to NIMA

- **CCE before irradiation**
 - 100 % with α particles and MIPS
 - tested on various samples 20-40 μ m
- **CCE after irradiation**
 - with α particles
 - neutron irradiated samples
 - material produced by CREE
 - 25 μ m thick layer



[S.Sciortino et al., presented on the RESMDD 04 conference, in press with NIMA]

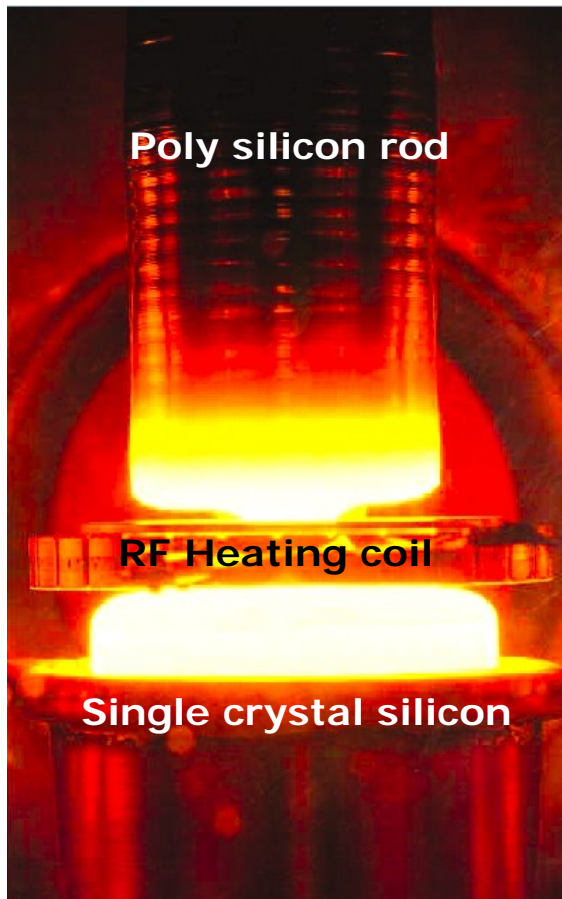
20% CCE (α)
after 7×10^{15} n/cm²!

35% CCE(β) (CCD $\sim 6\mu$ m ; ~ 300 e)
after 1.4×10^{16} p/cm²

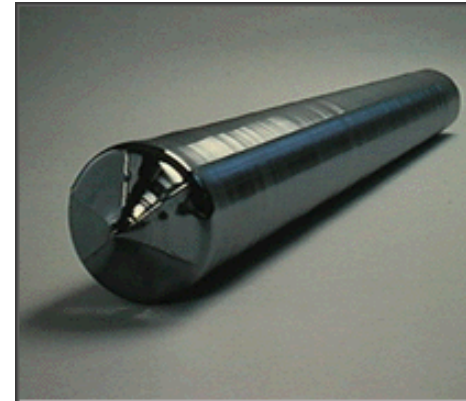
➔ much less than in silicon (see later)

■ Float Zone process

- Using a single Si crystal seed, melt the vertically oriented rod onto the seed using RF power and “pull” the **monocrystalline ingot**



■ Mono-crystalline Ingot



■ Wafer production

- Slicing, lapping, etching, polishing



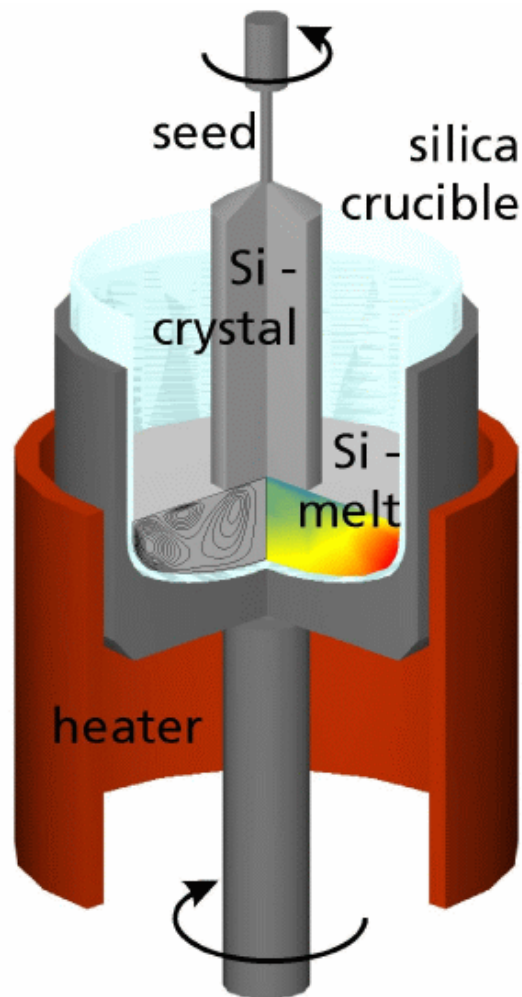
■ Oxygen enrichment (DOFZ)

- Oxidation of wafer at high temperatures

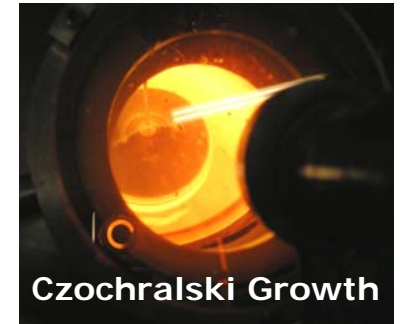
RD50 Czochralski silicon (Cz) & Epitaxial silicon (EPI)



■ Czochralski silicon



- Pull Si-crystal from a Si-melt contained in a silica crucible while rotating.
- Silica crucible is dissolving oxygen into the melt \Rightarrow **high concentration of O in CZ**
- Material used by IC industry (cheap)
- Recent developments (~2 years) made CZ available in sufficiently high purity (resistivity) to allow for use as particle detector.

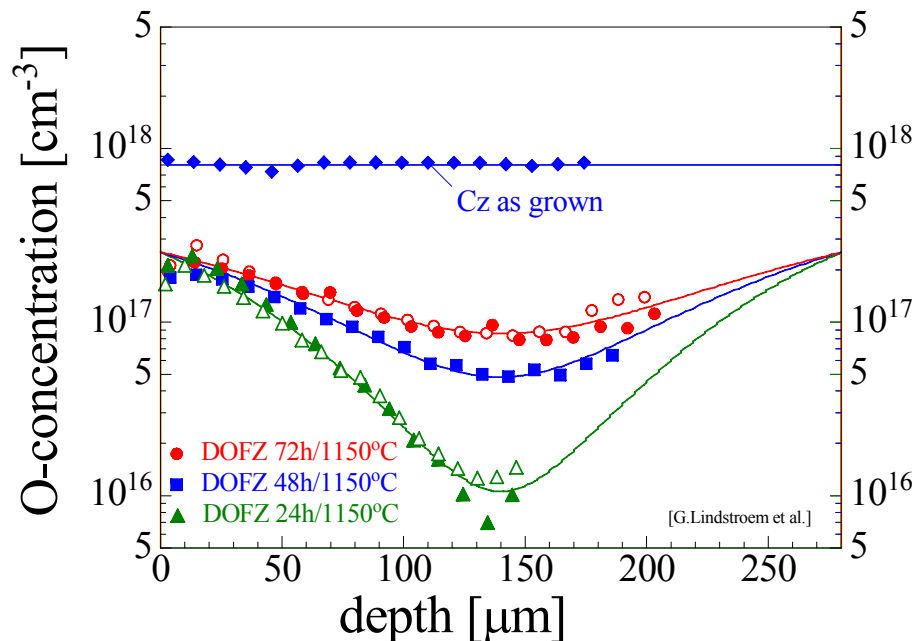


■ Epitaxial silicon

- Chemical-Vapor Deposition (CVD) of Silicon
- CZ silicon substrate used \Rightarrow **in-diffusion of oxygen**
- growth rate about $1\mu\text{m}/\text{min}$
- excellent homogeneity of resistivity
- up to $150\mu\text{m}$ thick layers produced (thicker is possible)
- price depending on thickness of epi-layer but not extending ~ 3 x price of FZ wafer

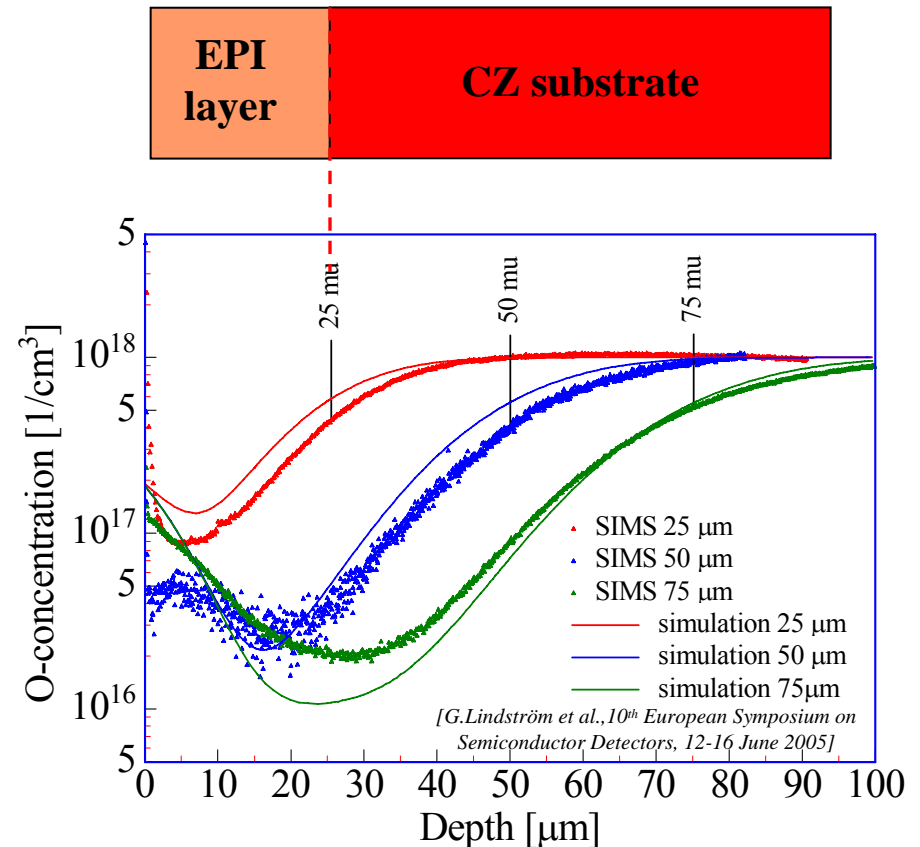
Cz and DOFZ silicon

- CZ: high O_i (oxygen) and O_{2i} (oxygen dimer) concentration (homogeneous)
- CZ: formation of Thermal Donors possible !



- DOFZ: inhomogeneous oxygen distribution
- DOFZ: oxygen content increasing with time at high temperature

Epitaxial silicon



- EPI: O_i and O_{2i} (?) diffusion from substrate into epi-layer during production
- EPI: in-homogeneous oxygen distribution

24 GeV/c proton irradiation

• Standard FZ silicon

- type inversion at $\sim 2 \times 10^{13}$ p/cm²
- strong N_{eff} increase at high fluence

• Oxygenated FZ (DOFZ)

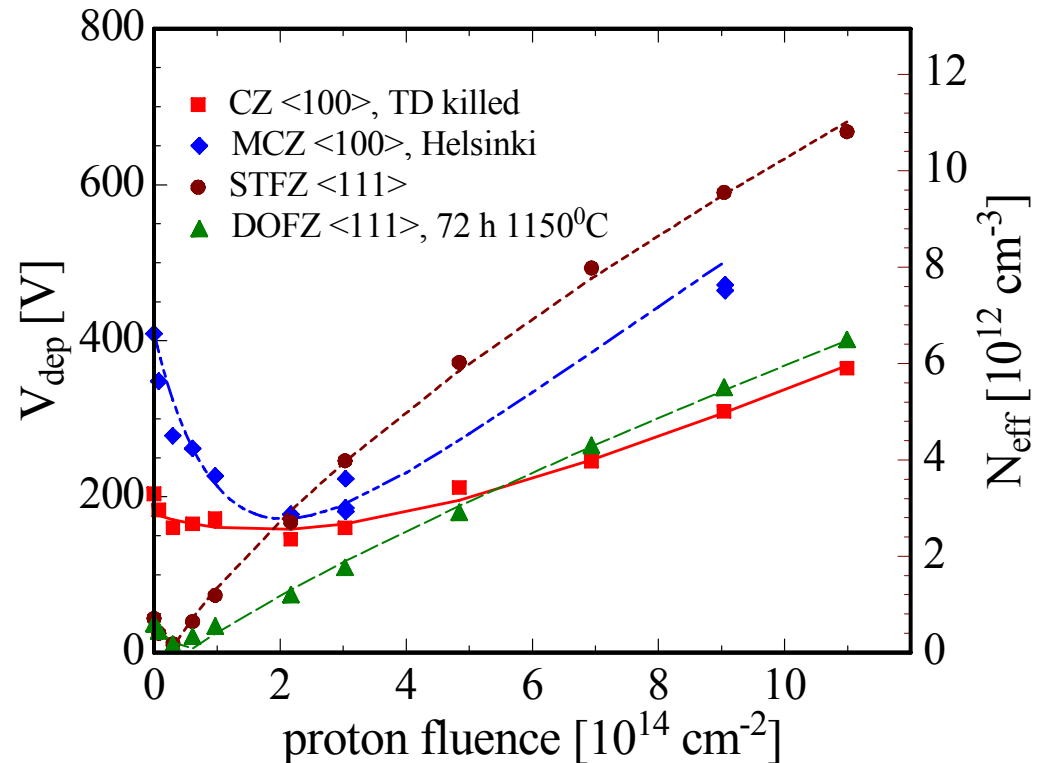
- type inversion at $\sim 2 \times 10^{13}$ p/cm²
- reduced N_{eff} increase at high fluence

• CZ silicon and MCZ silicon

- no type inversion in the overall fluence range (verified by TCT measurements)
(verified for CZ silicon by TCT measurements, preliminary result for MCZ silicon)
⇒ donor generation overcompensates acceptor generation in high fluence range

• Common to all materials (after hadron irradiation):

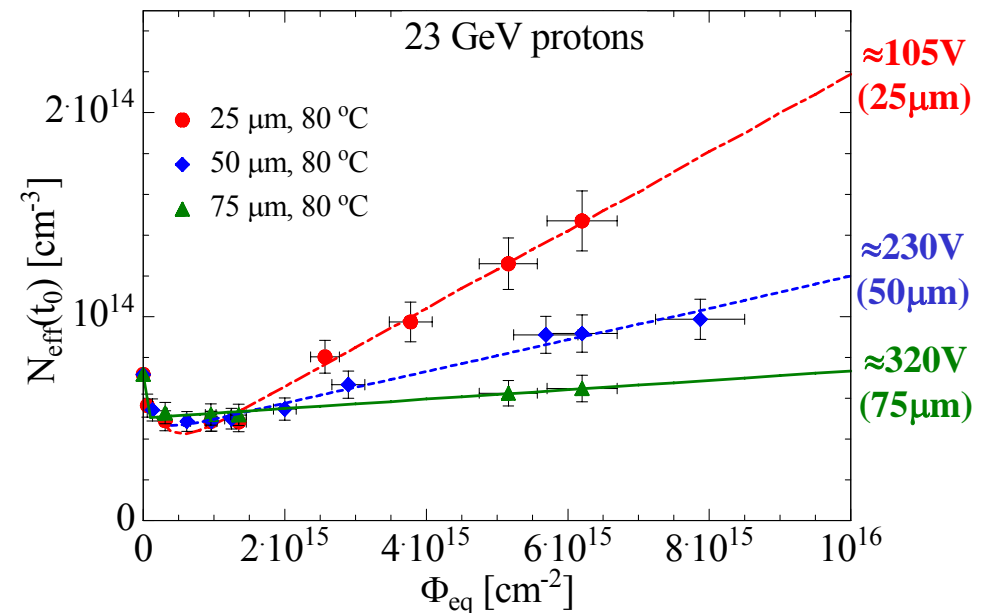
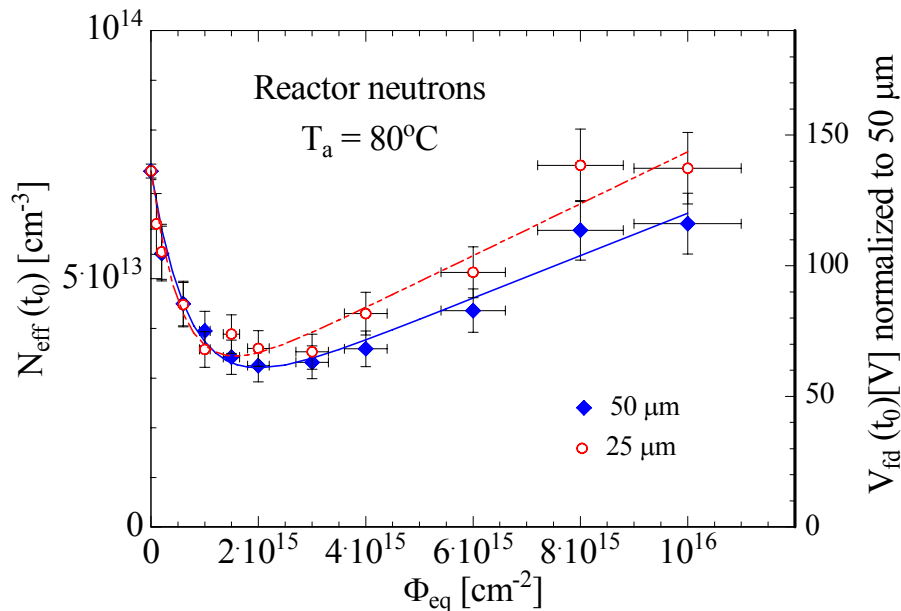
- reverse current increase
- increase of trapping (electrons and holes) within $\sim 20\%$



- **Epitaxial silicon grown by ITME**

- **Layer thickness: 25, 50, 75 μm ; resistivity: $\sim 50 \Omega\text{cm}$**
- **Oxygen: $[\text{O}] \approx 9 \times 10^{16} \text{cm}^{-3}$; Oxygen dimers (detected via IO_2 -defect formation)**

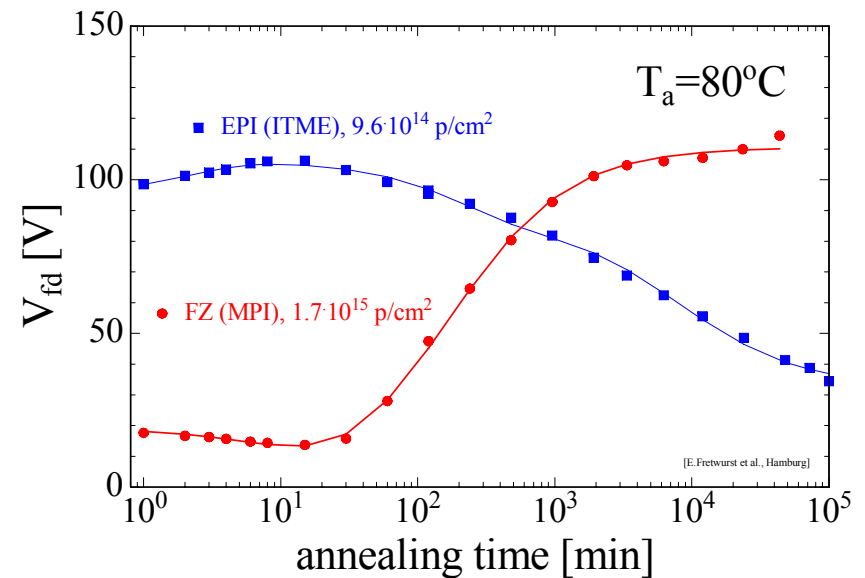
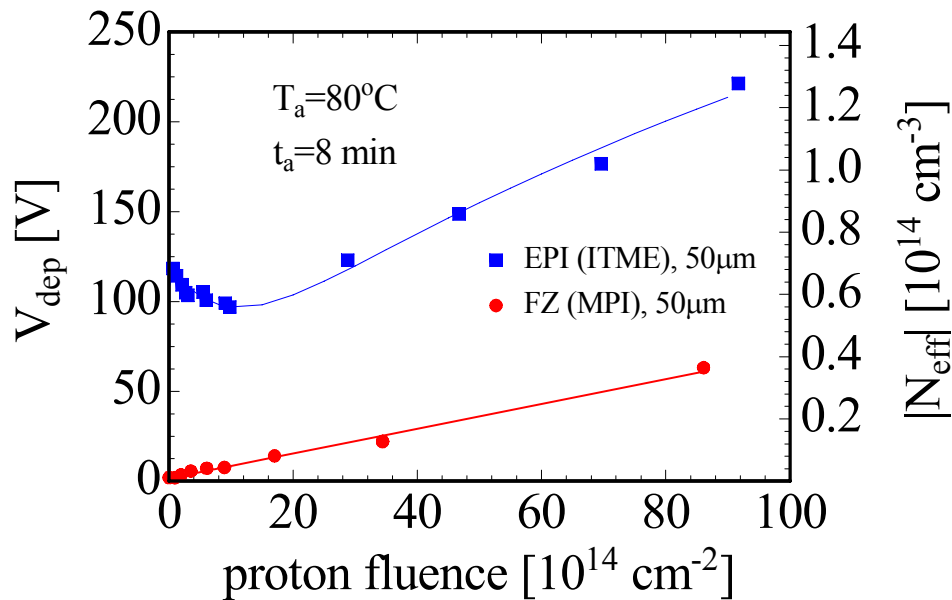
G.Lindström et al., 10th European Symposium on Semiconductor Detectors, 12-16 June 2005



- **No type inversion** in the full range up to $\sim 10^{16} \text{p/cm}^2$ and $\sim 10^{16} \text{n/cm}^2$ (type inversion only observed during long term annealing)

- **Proposed explanation:**
introduction of shallow donors bigger than generation of deep acceptors

- 50 μm thick silicon detectors:
 - **Epitaxial silicon** (50 Ωcm on CZ substrate, ITME & CiS)
 - **Thin FZ silicon** (4K Ωcm , MPI Munich, wafer bonding technique)



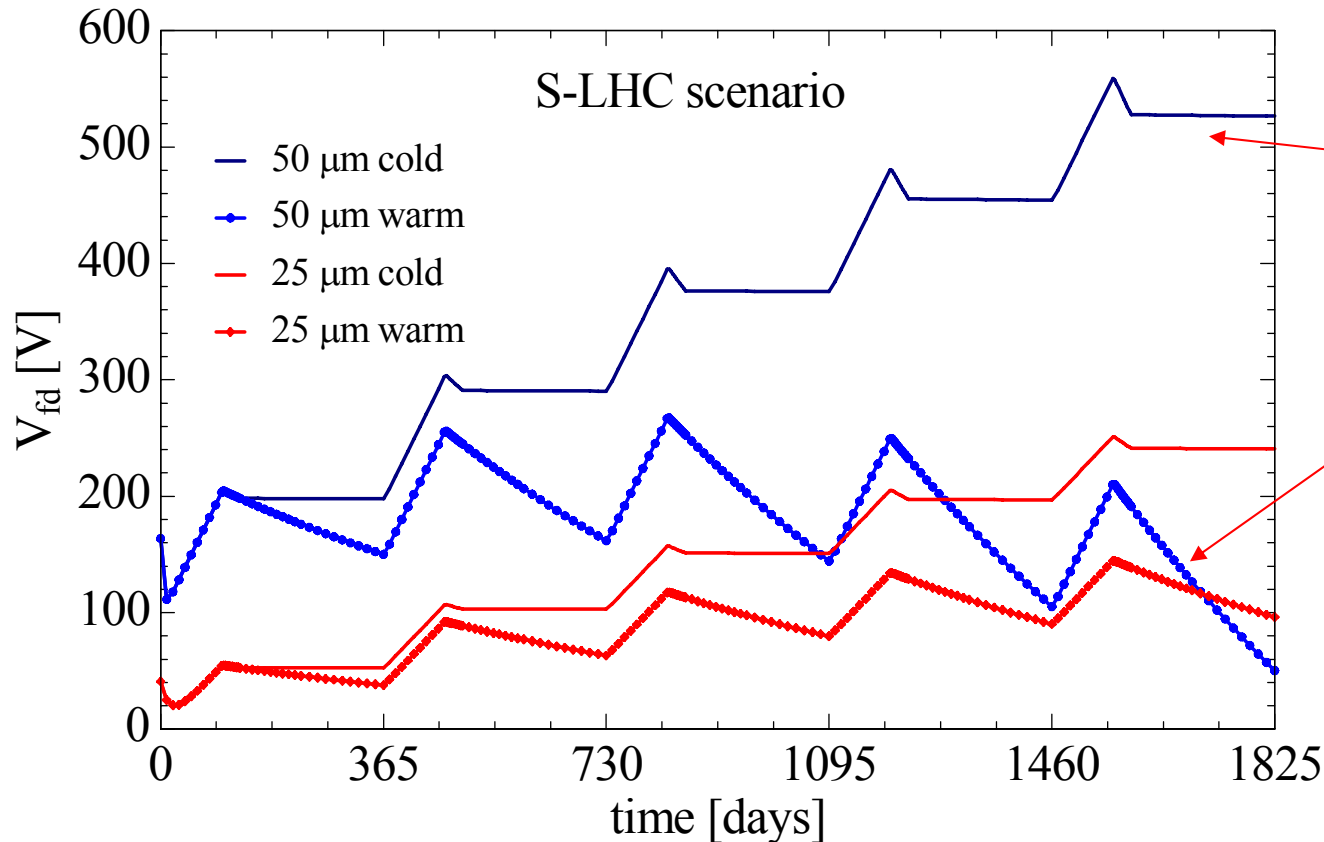
[E.Fretwurst et al., RESMDD - October 2004]

- **Thin FZ silicon:** Type inverted, increase of depletion voltage with time
- **Epitaxial silicon:** No type inversion, decrease of depletion voltage with time
 ⇒ No need for low temperature during maintenance of SLHC detectors!



G.Lindström et al., 10th European Symposium on Semiconductor Detectors, 12-16 June 2005 (Damage projection: M.Moll)

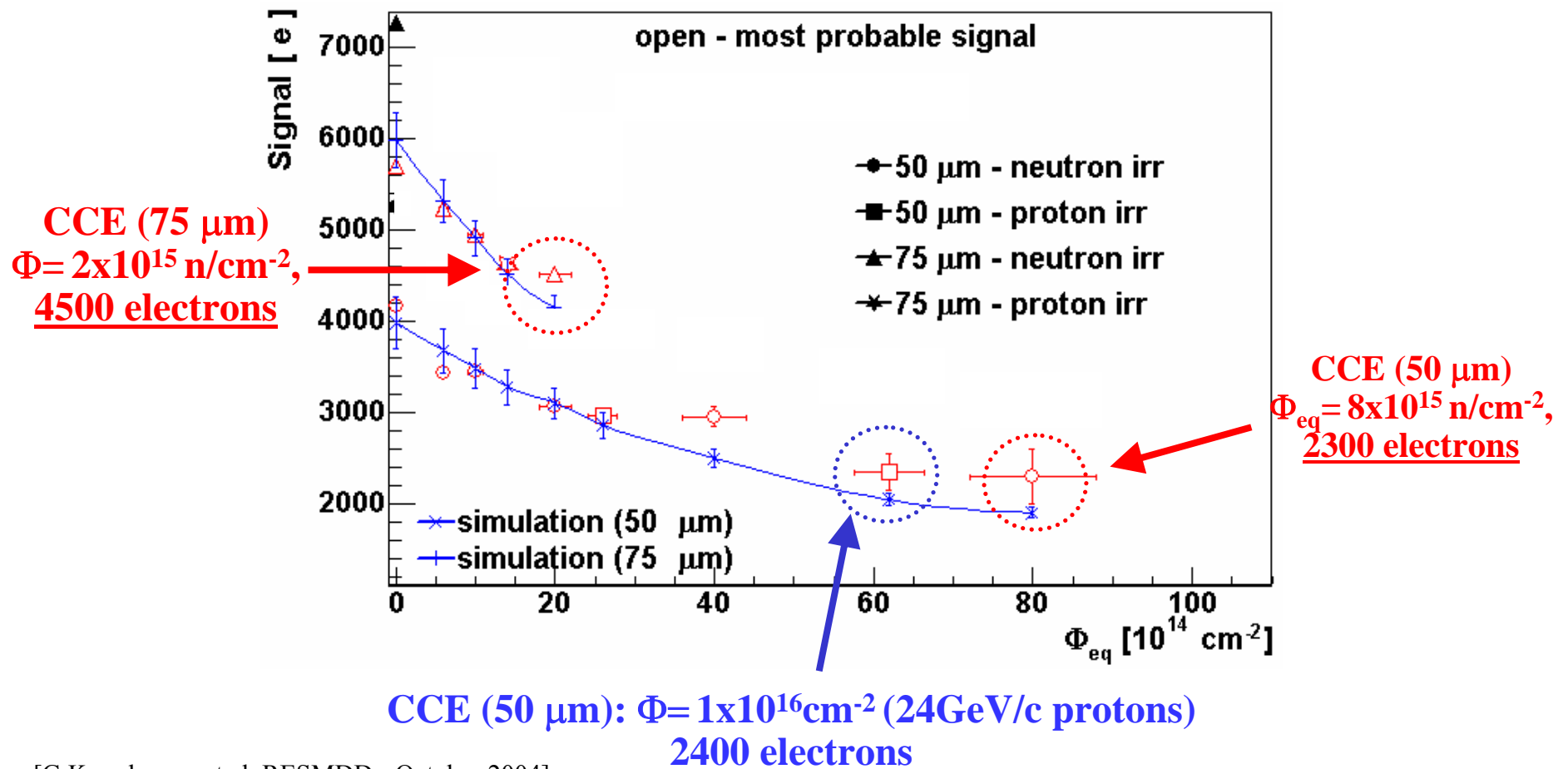
- **Radiation level (4cm):** $\Phi_{eq}(\text{year}) = 3.5 \times 10^{15} \text{ cm}^{-2}$
- **SLHC-scenario:** 1 year = 100 days beam (-7°C)
30 days maintenance (20°C)
235 days no beam (-7°C or 20°C)



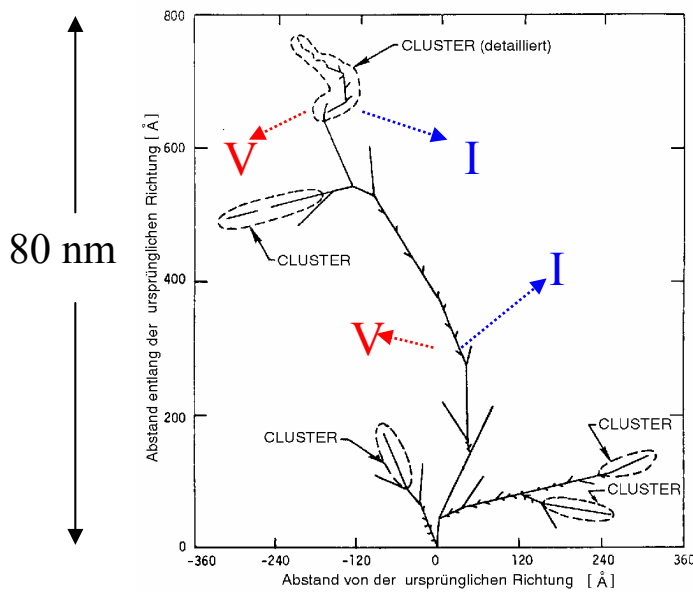
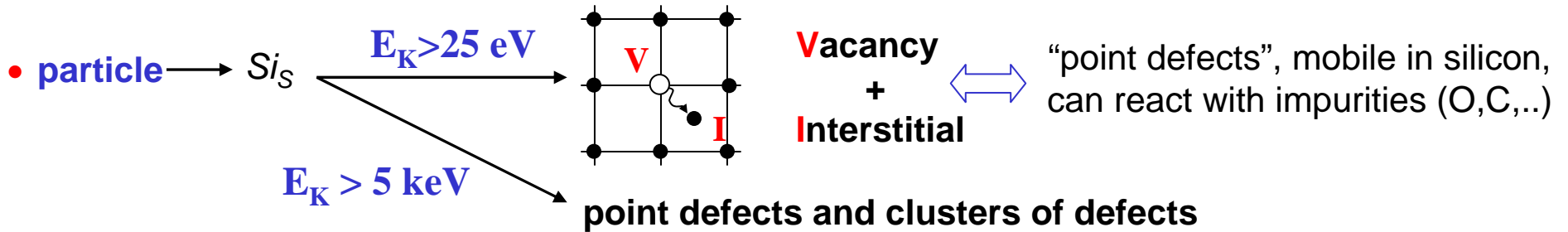
Detector with cooling when not operated

Detector without cooling when not operated

- Epitaxial silicon: CCE measured with beta particles (^{90}Sr)
 - 25ns shaping time
 - proton and neutron irradiations of 50 μm and 75 μm epi layers



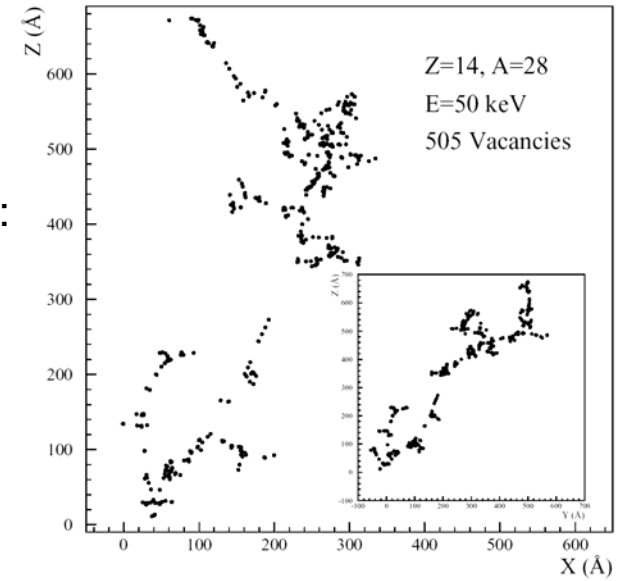
■ Damage to the silicon crystal: Displacement of lattice atoms



Distribution of vacancies created by a 50 keV Si-ion in silicon (typical recoil energy for 1 MeV neutrons):

← **Schematic** [Van Lint 1980]

Simulation → [M.Huhtinen 2001]



■ Defects can be electrically active (levels in the band gap)

- capture and release electrons and holes from conduction and valence band

⇒ can be charged - can be generation/recombination centers - can be trapping centers



- **2003:** Major breakthrough on γ -irradiated samples
 - For the first time macroscopic changes of the depletion voltage and leakage current can be explained by electrical properties of measured defects ! [APL, 82, 2169, March 2003]
- **2004:** Big step in understanding the improved radiation tolerance of oxygen enriched and epitaxial silicon after proton irradiation

[I.Pintilie, RESMDD, Oct.2004]

Levels responsible for depletion voltage changes after proton irradiation:

Almost independent of oxygen content:

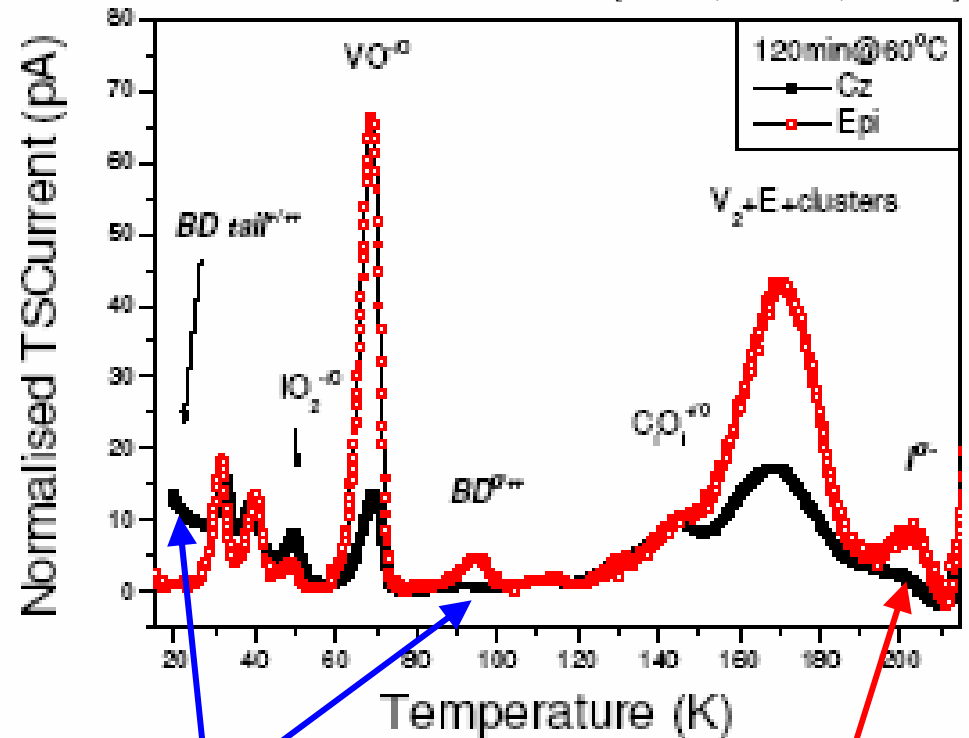
- Donor removal
- “Cluster damage” \Rightarrow negative charge

Influenced by initial oxygen content:

- **I-defect:** deep acceptor level at $E_C - 0.54\text{eV}$ (good candidate for the V_2O defect) \Rightarrow negative charge

Influenced by initial oxygen dimer content (?):

- **BD-defect:** bistable shallow thermal donor (formed via oxygen dimers O_{2i}) \Rightarrow positive charge



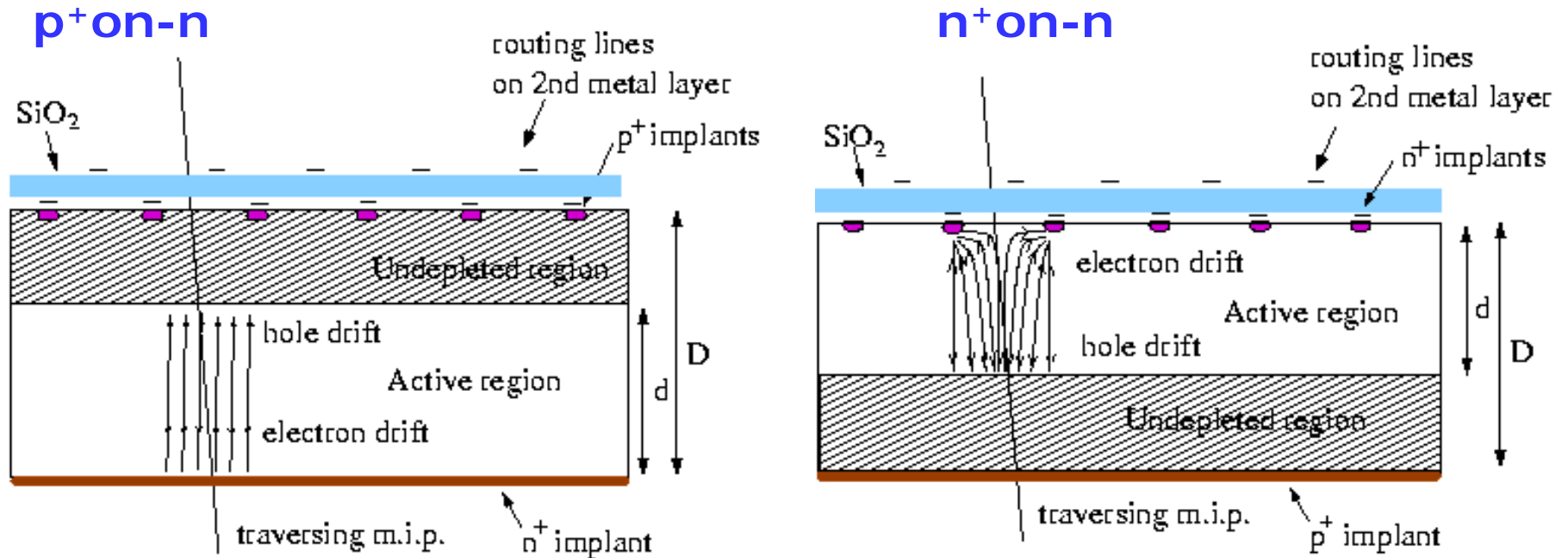
BD-defect

I-defect



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n-type silicon after type inversion:



p-on-n silicon, under-depleted:

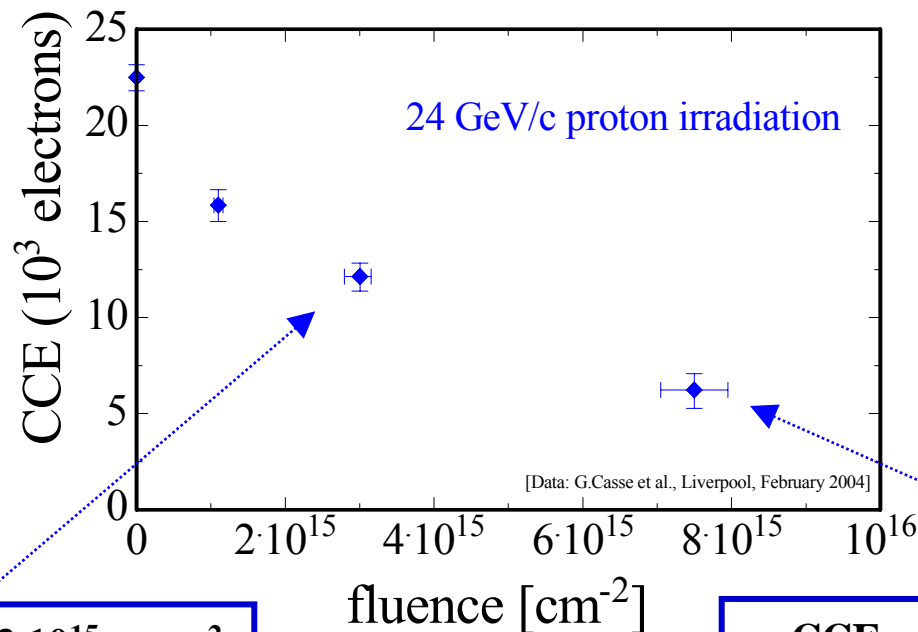
- Charge spread – degraded resolution
- Charge loss – reduced CCE

n-on-n silicon, under-depleted:

- Limited loss in CCE
- Less degradation with under-depletion
- Collect electrons (fast)

n-in-p: - no type inversion, high electric field stays on structured side
- collection of electrons

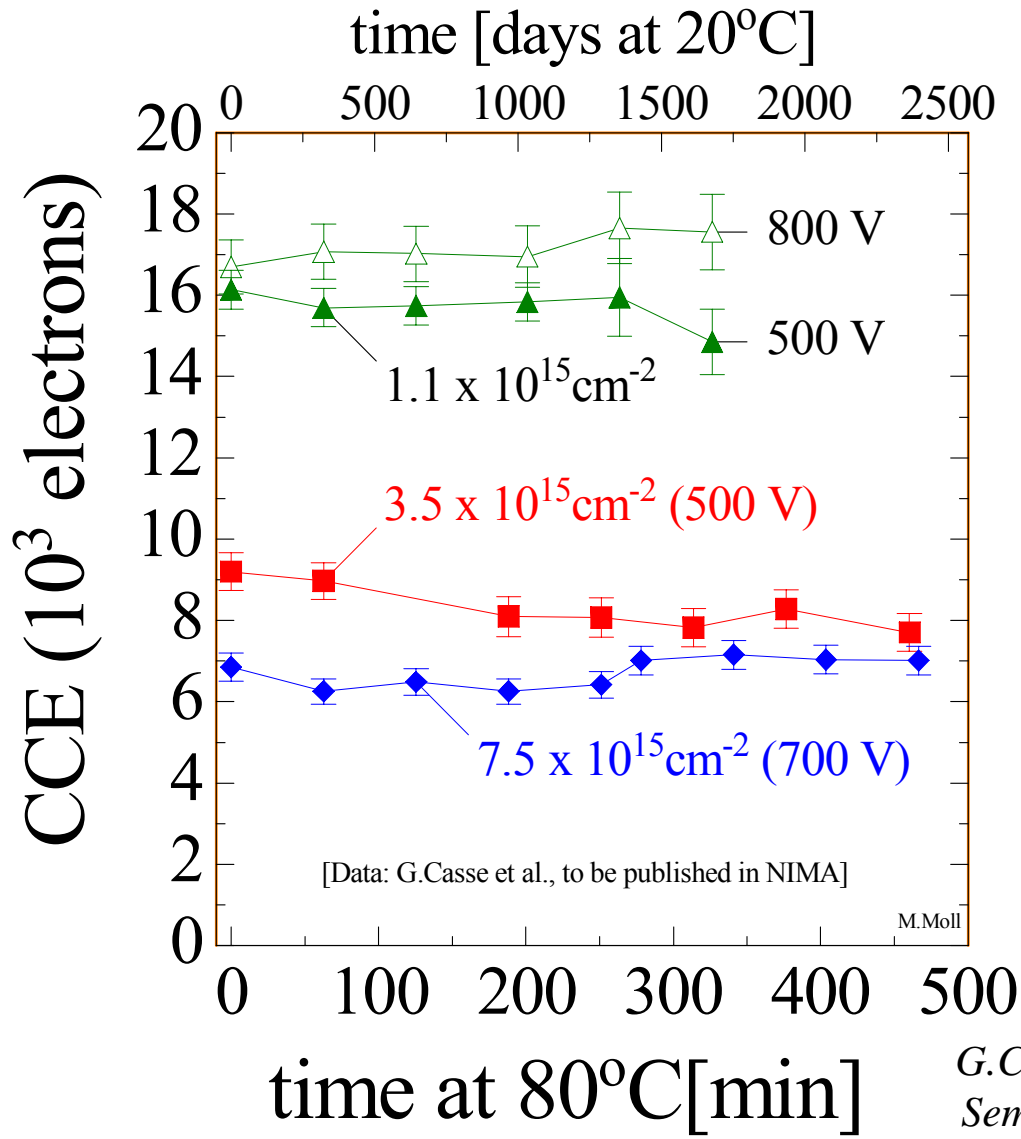
- Miniature n-in-p microstrip detectors (280 μ m)
- Detectors read-out with LHC speed (40MHz) chip (SCT128A)
- Material: standard p-type and oxygenated (DOFZ) p-type
- Irradiation:



At the highest fluence
Q~6500e at $V_{\text{bias}}=900\text{V}$

**CCE ~ 60% after $3 \times 10^{15} \text{ p cm}^{-2}$
at 900V (standard p-type)**

**CCE ~ 30% after $7.5 \times 10^{15} \text{ p cm}^{-2}$
900V (oxygenated p-type)**



- p-type strip detector ($280 \mu\text{m}$) irradiated with 23 GeV p ($7.5 \times 10^{15} \text{ p/cm}^2$)

- expected from previous CV measurement of V_{dep} :
- before reverse annealing:

$$V_{\text{dep}} \sim 2800 \text{ V}$$

- after reverse annealing

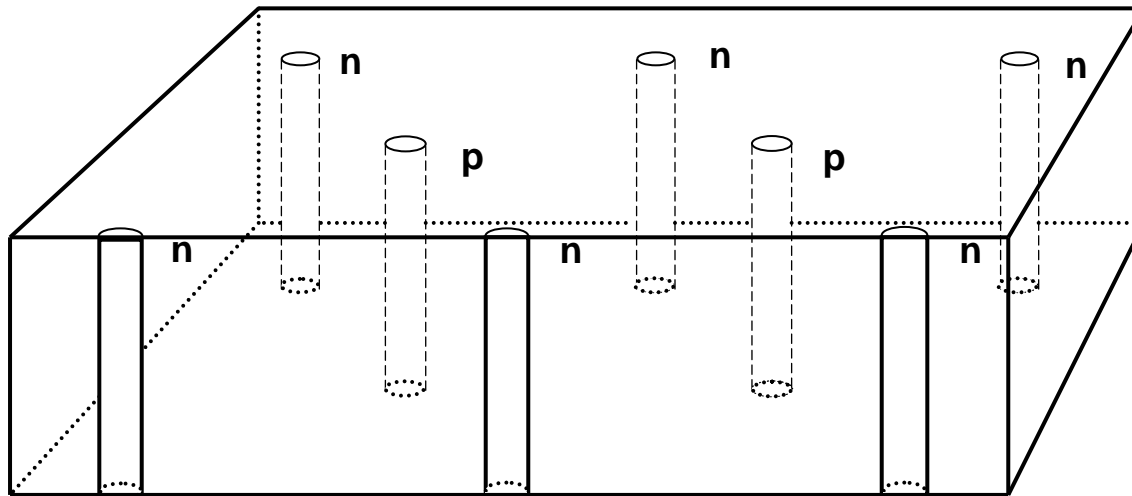
$$V_{\text{dep}} > 12000 \text{ V}$$

- no reverse annealing visible in the CCE measurement !

G.Casse et al., 10th European Symposium on Semiconductor Detectors, 12-16 June 2005

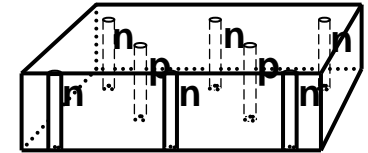
- **Introduced by:** - S.I. Parker C.J. Kenney and J. Segal, NIMA 395 (1997) 328
- **“3D” electrodes:** - narrow columns along detector thickness,
- diameter: $10\mu\text{m}$, distance: $50 - 100\mu\text{m}$
- **Lateral depletion:** - lower depletion voltage needed
- thicker detectors possible
- fast signal

See presentation on Thursday Morning:
S.Parker “3D sensors”





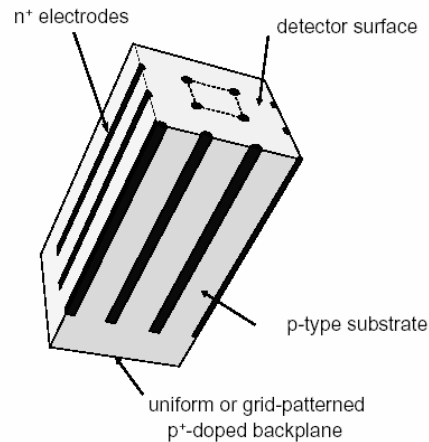
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• Simplified 3D architecture

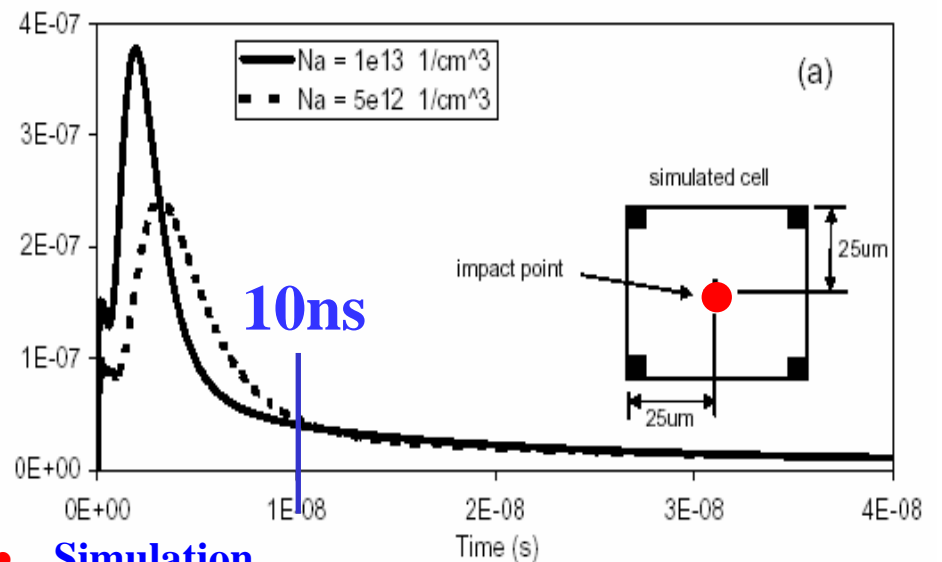
- n^+ columns in p-type substrate, p^+ backplane
- operation similar to standard 3D detector



[C. Piemonte et al.,
NIM A541 (2005) 441]

• Simplified process

- hole etching and doping only done once
- no wafer bonding technology needed



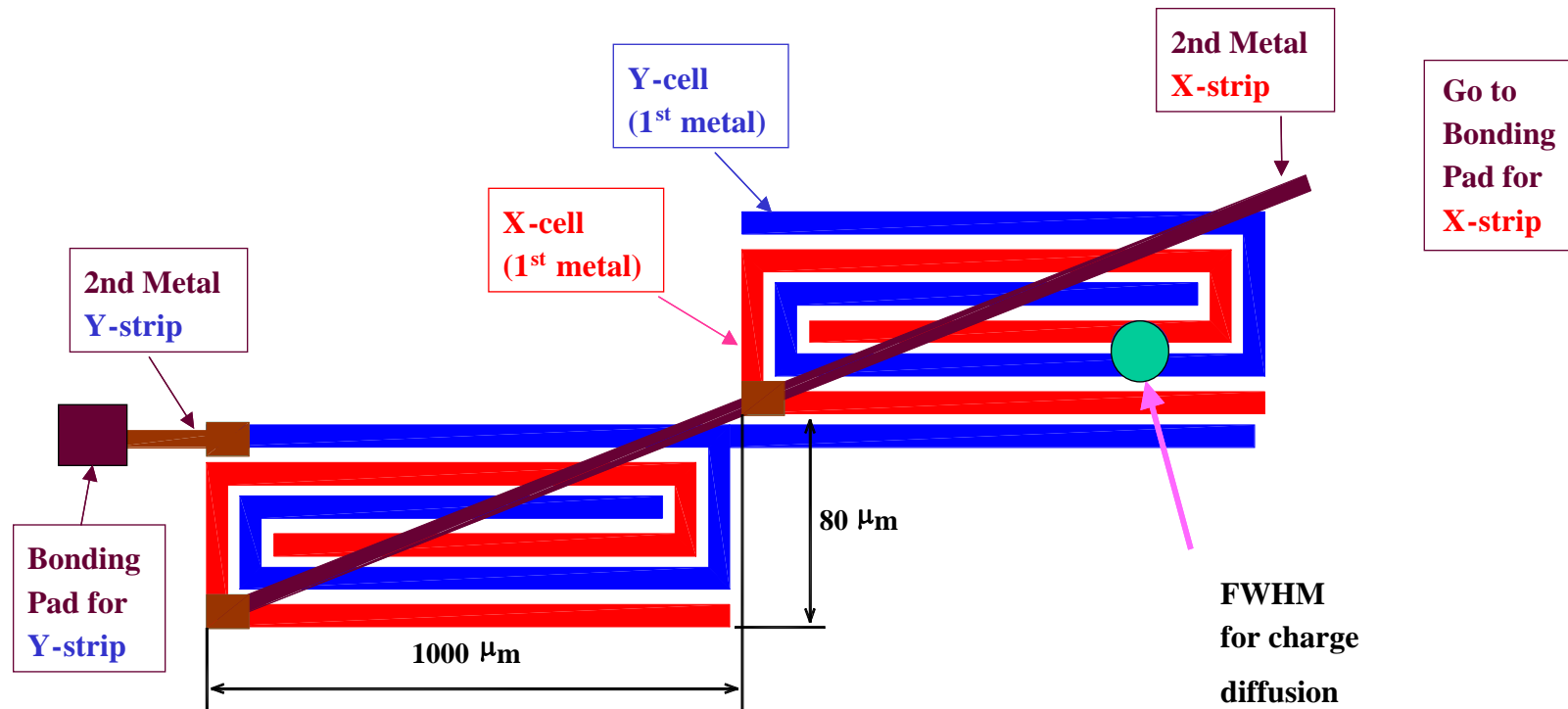
• Simulation

- worst case shown (hit in middle of cell)
- still CCE below 10ns possible

• Fabrication:

- under way IRST(Italy), CNM Barcelona

- **New structures:** There is a multitude of concepts for new (planar and mixed planar & 3D) detector structures aiming for improved radiation tolerance or less costly detectors (see e.g. Z.Li - 6th RD50 workshop)
- **Example: Stripixel concept:**



Z. Li, D. Lissauer, D. Lynn, P. O'Connor, V. Radeka



- **At fluences up to 10^{15}cm^{-2} (Outer layers of a SLHC detector) the change of depletion voltage and the large area to be covered by detectors is the major problem.**
 - **CZ silicon detectors** could be a cost-effective radiation hard solution
(no type inversion, use p-in-n technology)
 - **p-type silicon microstrip detectors show very encouraging results:**
CCE ≈ 6500 e; $\Phi_{\text{eq}} = 4 \times 10^{15} \text{ cm}^{-2}$, $300\mu\text{m}$, collection of electrons,
no reverse annealing observed in CCE measurement!
- **At the fluence of 10^{16}cm^{-2} (Innermost layer of a SLHC detector) the active thickness of any silicon material is significantly reduced due to trapping.**
The promising new options are:
 - Thin/EPI detectors : drawback: radiation hard electronics for low signals needed**
e.g. 2300e at $\Phi_{\text{eq}} 8 \times 10^{15}\text{cm}^{-2}$, $50\mu\text{m}$ EPI,
.... thicker layers will be tested in 2005/2006
 - 3D detectors : drawback: technology has to be optimized**
..... steady progress within RD50
- **New Materials like SiC and GaN (not shown) have been characterized .**
CCE tests show that these materials are not radiation harder than silicon

Further information: <http://cern.ch/rd50/>