

**Recent Developments on
Radiation Hard Semiconductor Detectors for SuperLHC
- CERN-RD50 project -**

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on behalf of RD50
CERN - Geneva - Switzerland

OUTLINE

- **Motivation to develop radiation harder detectors**
- **The RD50 collaboration**
- **Radiation Damage – a very brief review**
- **Approaches to obtain radiation hard sensors**
 - **Material engineering**
 - **Device engineering**
- **Summary**

<http://www.cern.ch/rd50>

- **LHC upgrade** (“Super-LHC” ... later than 2010)

LHC: $L = 10^{34} \text{cm}^{-2} \text{s}^{-1}$ $\xrightarrow{10 \text{ years}}$ $\phi(\text{R}=4\text{cm}) \sim 3 \cdot 10^{15} \text{cm}^{-2}$
 $\phi(\text{R}=75\text{cm}) \sim 3 \cdot 10^{13} \text{cm}^{-2}$

⇒ Technology available ⇒ However, serious radiation damage!

S-LHC: $L = 10^{35} \text{cm}^{-2} \text{s}^{-1}$ $\xrightarrow{5 \text{ years}}$ $\phi(\text{R}=4\text{cm}) \sim 1.6 \cdot 10^{16} \text{cm}^{-2}$

⇒ Technology not available ⇒ Focused and coordinated R&D mandatory to develop radiation hard and cost-effective detectors

- **LHC experiments** (...starting 2007)

⇒ **Radiation hard technologies now adopted have not been completely characterized:** Oxygen-enriched Si in ATLAS/CMS pixels

⇒ **Replacement of components** e.g. for LHCb Velo at $r < 4\text{cm}$ a replacement of detectors is foreseen after 3 years operation

- **Linear collider experiments**

Deep understanding of radiation damage will be fruitful for linear collider experiments where high doses of e, γ will play a significant role.

RD50: Development of Radiation Hard Semiconductor Devices for High Luminosity Colliders

- Collaboration formed in November 2001
- Experiment approved as RD50 by CERN in June 2002
- Main objective:

Development of ultra-radiation hard semiconductor detectors for the luminosity upgrade of the LHC to $10^{35} \text{ cm}^{-2}\text{s}^{-1}$ (“Super-LHC”).

Challenges:

- Radiation hardness up to 10^{16} cm^{-2} required
- Fast signal collection (Going from 25ns to 10 ns bunch crossing ?)
- Low mass (reducing multiple scattering close to interaction point)
- Cost effectiveness (big surfaces have to be covered with detectors!)

- Presently 271 Members from 52 Institutes

Belarus (Minsk), **Belgium** (Louvain), **Canada** (Montreal), **Czech Republic** (Prague (2x)), **Finland** (Helsinki, Lappeenranta), **Germany** (Berlin, Dortmund, Erfurt, Hamburg, Karlsruhe), **Israel** (Tel Aviv), **Italy** (Bari, Bologna, Florence, Milano, Padova, Perugia, Pisa, Trento, Trieste, Turin), **Lithuania** (Vilnius), **Norway** (Oslo (2x)), **Poland** (Warsaw), **Romania** (Bucharest (2x)), **Russia** (Moscow (2x), St.Petersburg), **Slovenia** (Ljubljana), **Spain** (Barcelona, Valencia), **Switzerland** (CERN, PSI), **Ukraine** (Kiev), **United Kingdom** (Exeter, Glasgow, Lancaster, Liverpool, Sheffield, University of Surrey), **USA** (Fermilab, Purdue University, Rochester University, Rutgers University, SCIPP Santa Cruz, Syracuse University, BNL, University of New Mexico)

Scientific strategies:

- I. Material engineering
- II. Device engineering
- III. Variation of detector operational conditions

CERN-RD39

“Cryogenic Tracking Detectors”

<http://cern.ch/rd39>

• Defect Engineering of Silicon

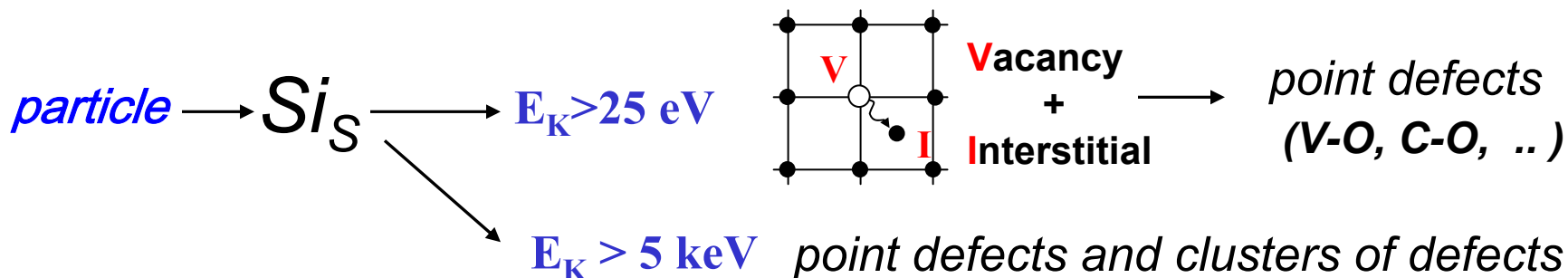
- Understanding radiation damage
 - Macroscopic effects and Microscopic defects
 - *Simulation of defect properties and defect kinetics*
 - *Irradiation with different type of particles at different energies*
- Oxygen enriched silicon
 - DOFZ – Diffusion Oxygenated Float Zone Silicon
 - Cz - Czochralski Silicon
 - MCZ - Magnetic Czochralski
 - EPI – Epitaxial silicon grown on CZ substrate
- *Oxygen dimer enriched silicon*
- *Hydrogen enriched silicon*
- *Pre-irradiated silicon*

• New Materials

- Silicon Carbide (SiC)
- Gallium Nitride (GaN)
- *(Diamond: CERN RD42 Collaboration)*

• Device Engineering (New Detector Designs)

- Thin detectors
- 3D detectors
- Semi 3D detectors
- *Cost effective detectors*
- p-type silicon detectors
- *Simulation of highly irradiated detectors*



• ^{60}Co -gammas

- Compton Electrons with max. $E_\gamma \approx 1 \text{ MeV}$ (no cluster production)

• Electrons

- $E_e > 255 \text{ keV}$ for displacement
- $E_e > 8 \text{ MeV}$ for cluster

• Neutrons (elastic scattering)

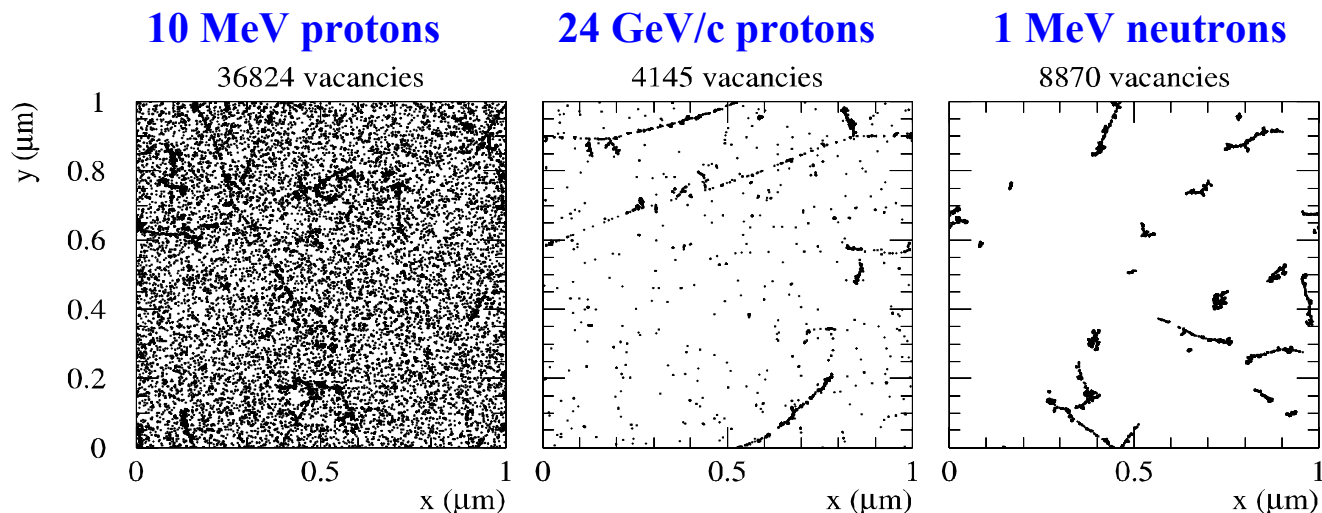
- $E_n > 185 \text{ eV}$ for displacement
- $E_n > 35 \text{ keV}$ for cluster

Only point defects \longleftrightarrow **point defects & clusters** \longleftrightarrow **Mainly clusters**

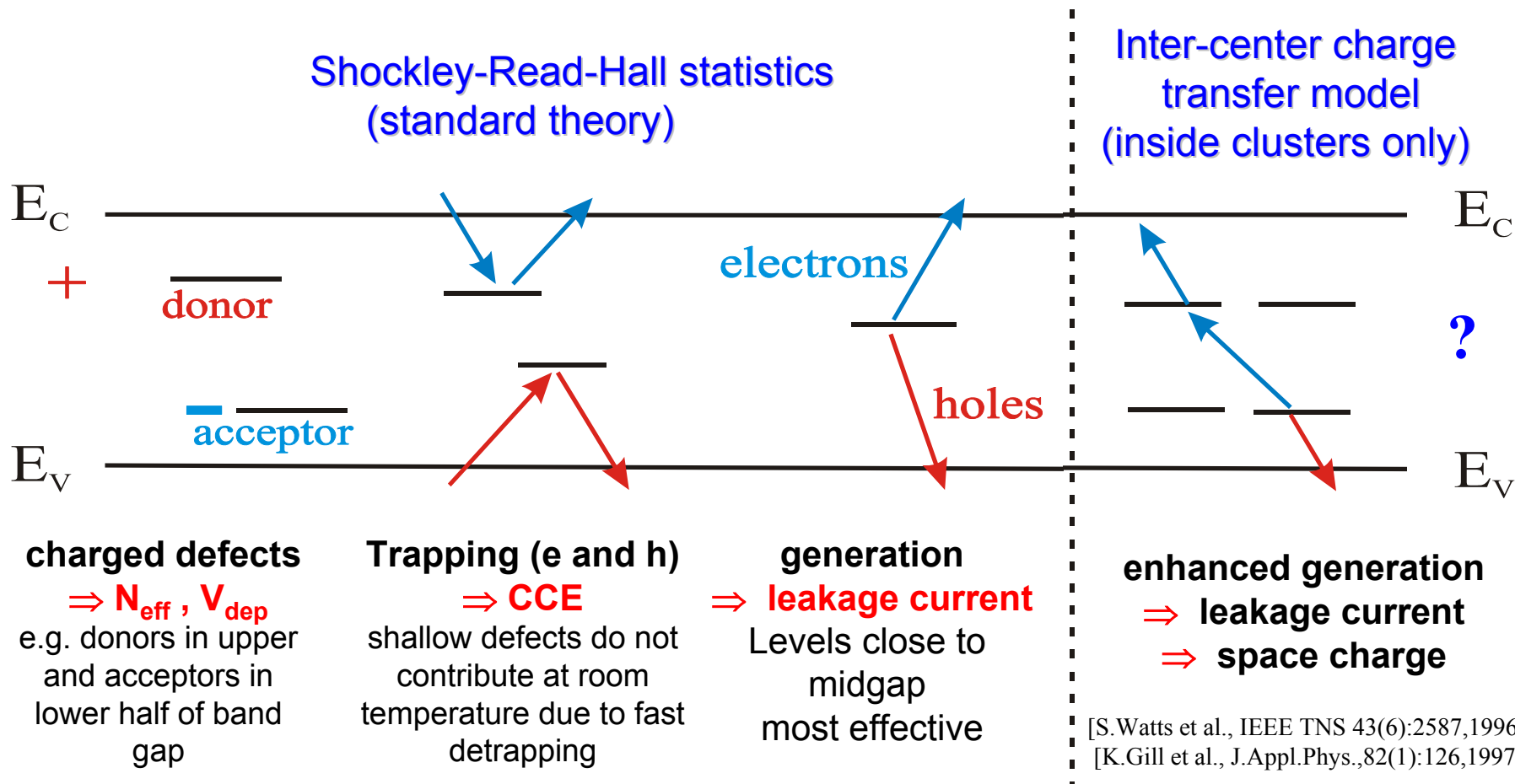
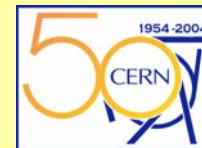
Simulation:

Initial distribution of vacancies in $(1\mu\text{m})^3$ after 10^{14} particles/cm 2

[Mika Huhtinen NIMA 491(2002) 194]



RD50 Impact of Defects on Detector properties



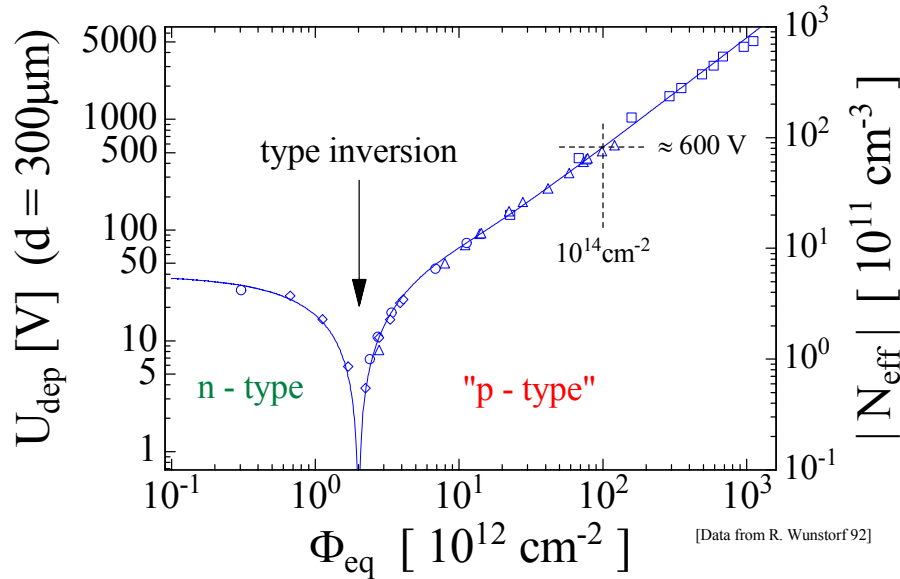
Impact on detector properties can be calculated if all defect parameters are known:

$\sigma_{n,p}$: cross sections

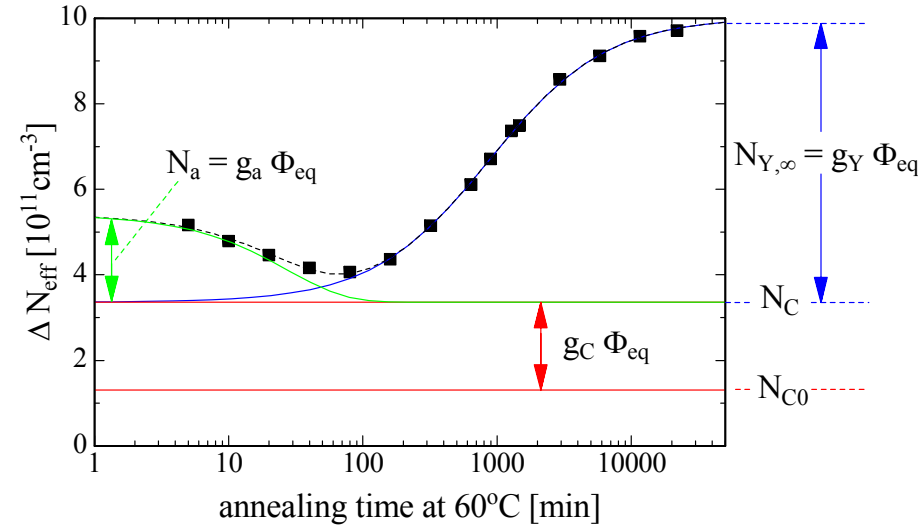
ΔE : ionization energy

N_t : concentration

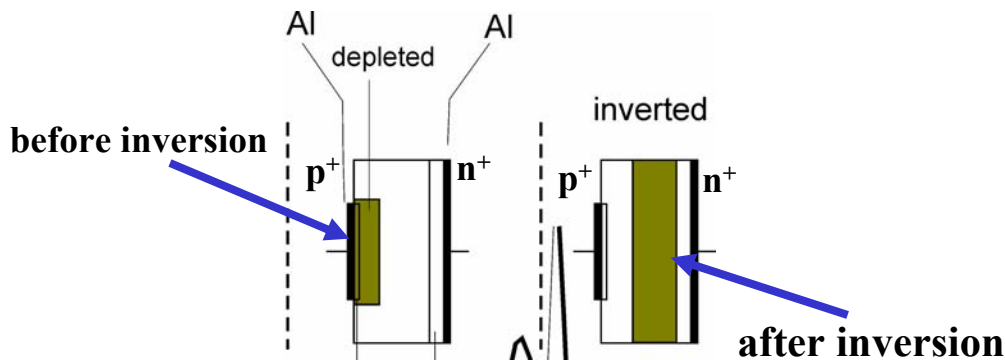
Change of V_{dep} (N_{eff})



Annealing



- Type inversion:
SCSI – Space Charge Sign Inversion

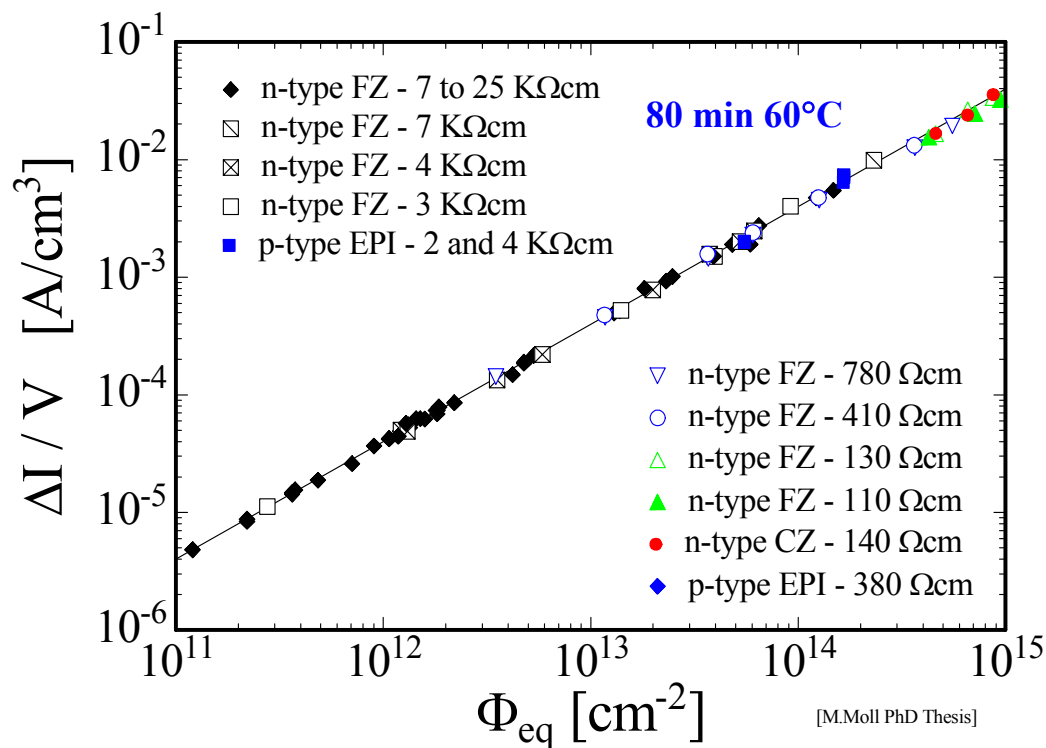


- Short term: **“Beneficial annealing”**
- Long term: **“Reverse annealing”**

time constant :

- ~ 500 years (-10°C)
- ~ 500 days (20°C)
- ~ 21 hours (60°C)

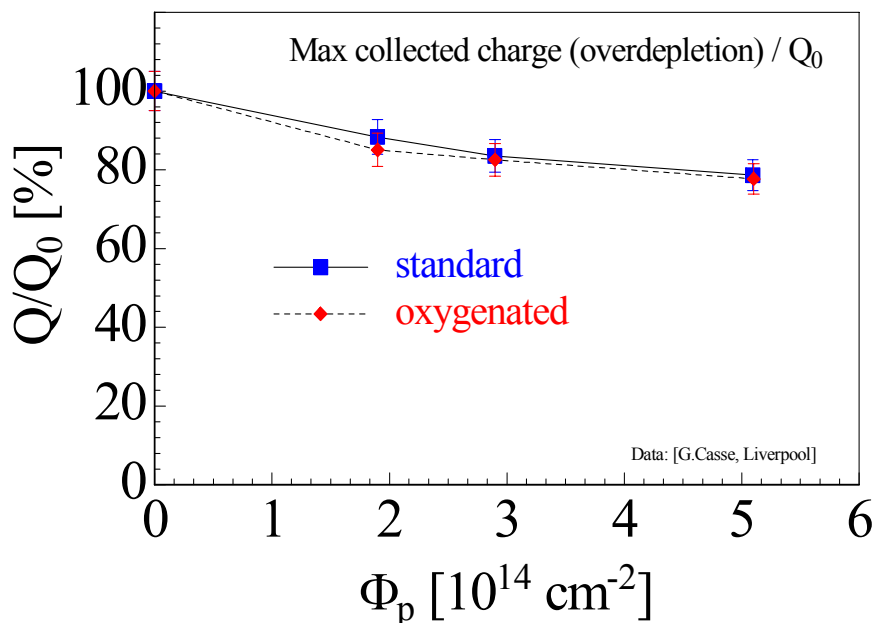
Hadron irradiation



- Damage parameter α (slope) independent of fluence
- α independent of Φ_{eq} and impurities
 - ⇒ can be used for fluence calibration (NIEL-Hypothesis)
(take care about annealing effects!)

- **Mechanisms reducing collected charge:**
 - **Trapping (electrons and holes)**
 - **Underdepletion (detector design and geometry)**
 - **Type inversion**

ATLAS microstrip + RO electronics



- **Oxygenation has no influence on trapping.**
- **After $5 \cdot 10^{14} \text{ p/cm}^2$ (24 GeV/c)**
 - 80% of charge collected (25ns)
 - overdepletion needed !

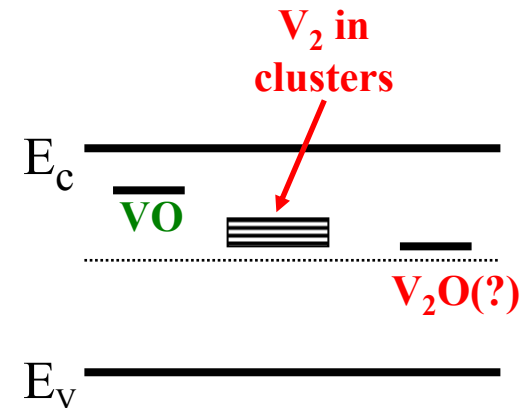
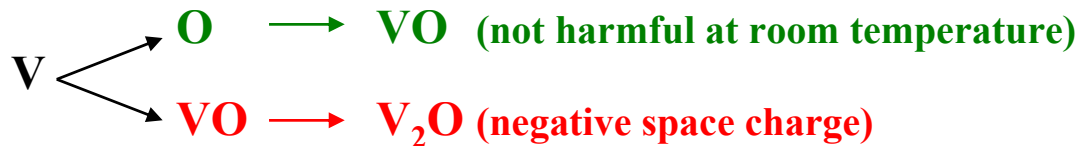
Data: Gianluigi Casse; 1st Workshop on Radiation Hard Semiconductor Devices for High Luminosity Colliders; CERN; 28-30 November 2002

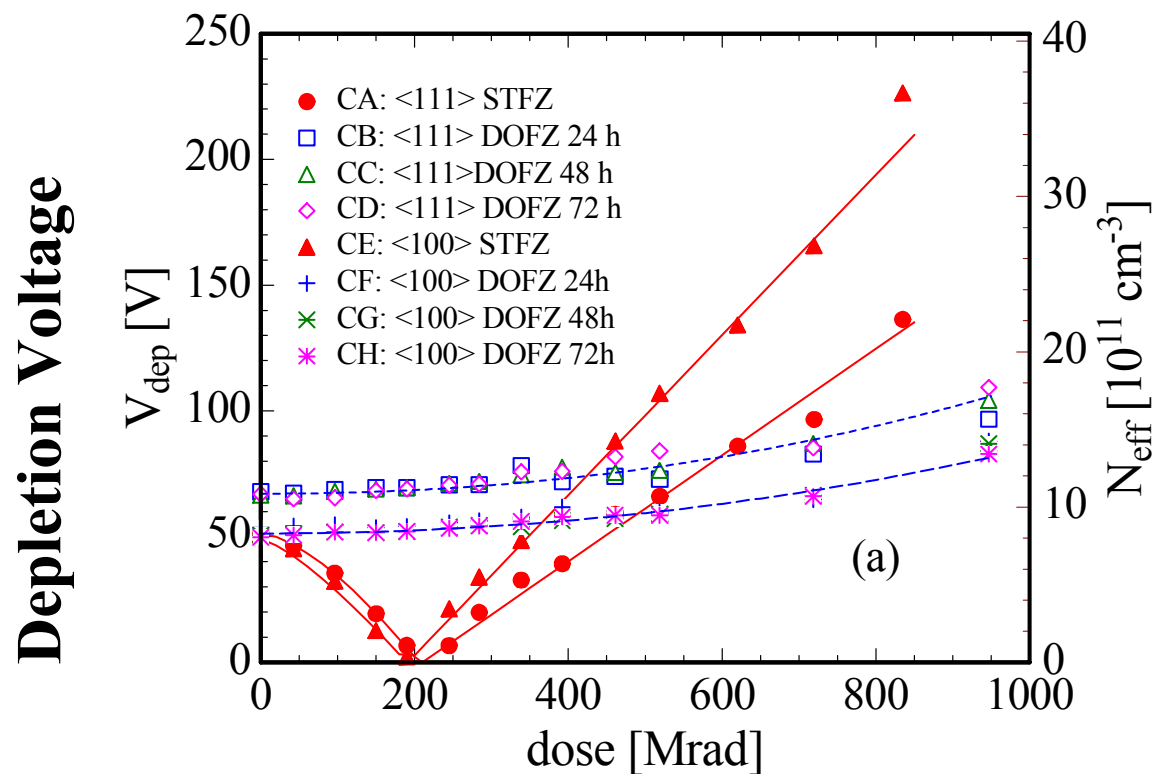
- Influence the defect kinetics by incorporation of impurities or defects
- Best example: Oxygen

Initial idea: Incorporate Oxygen to getter radiation-induced vacancies
 \Rightarrow prevent formation of Di-vacancy (V_2) related deep acceptor levels

Observation: Higher oxygen content \Rightarrow less negative space charge
 (less charged acceptors)

- One possible mechanism: V_2O is a deep acceptor





[E.Fretwurst et al. 1st RD50 Workshop]

See also:

- Z.Li et al. [NIMA461(2001)126]

- Z.Li et al. [1st RD50 Workshop]

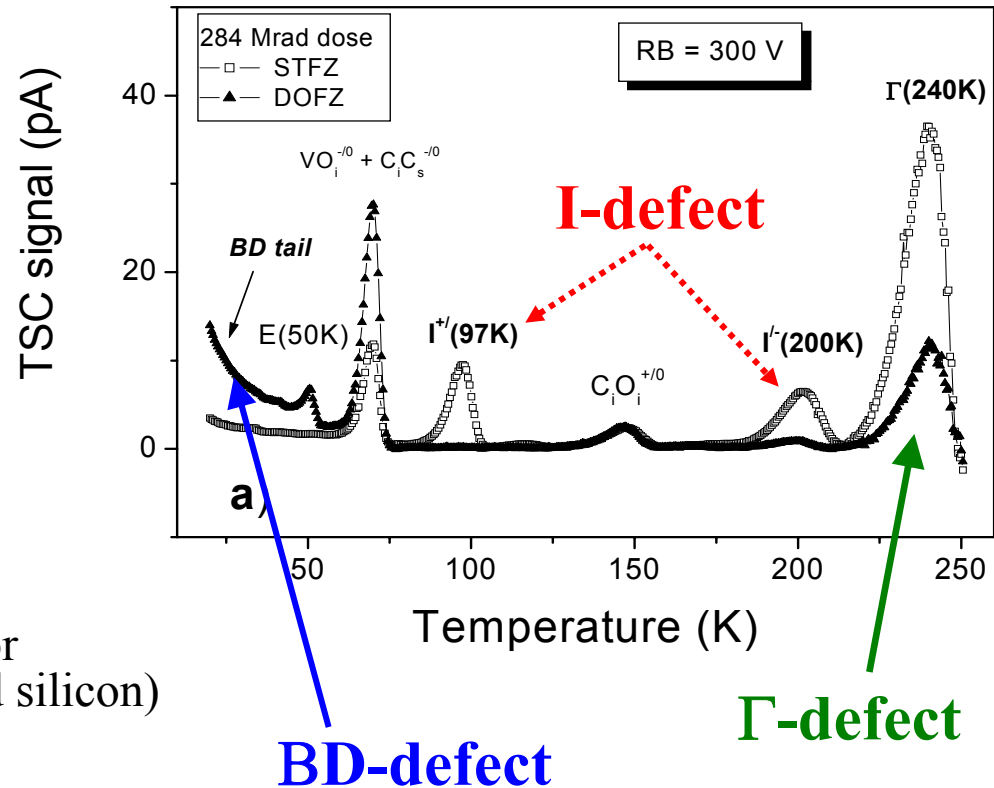
- **No type inversion for oxygen enriched silicon!**
- **Slight increase of positive space charge**
(due to Thermal Donor generation?)

- **For the first time** macroscopic changes of the depletion voltage and leakage current can be explained by electrical properties of measured defects → **Major breakthrough!**

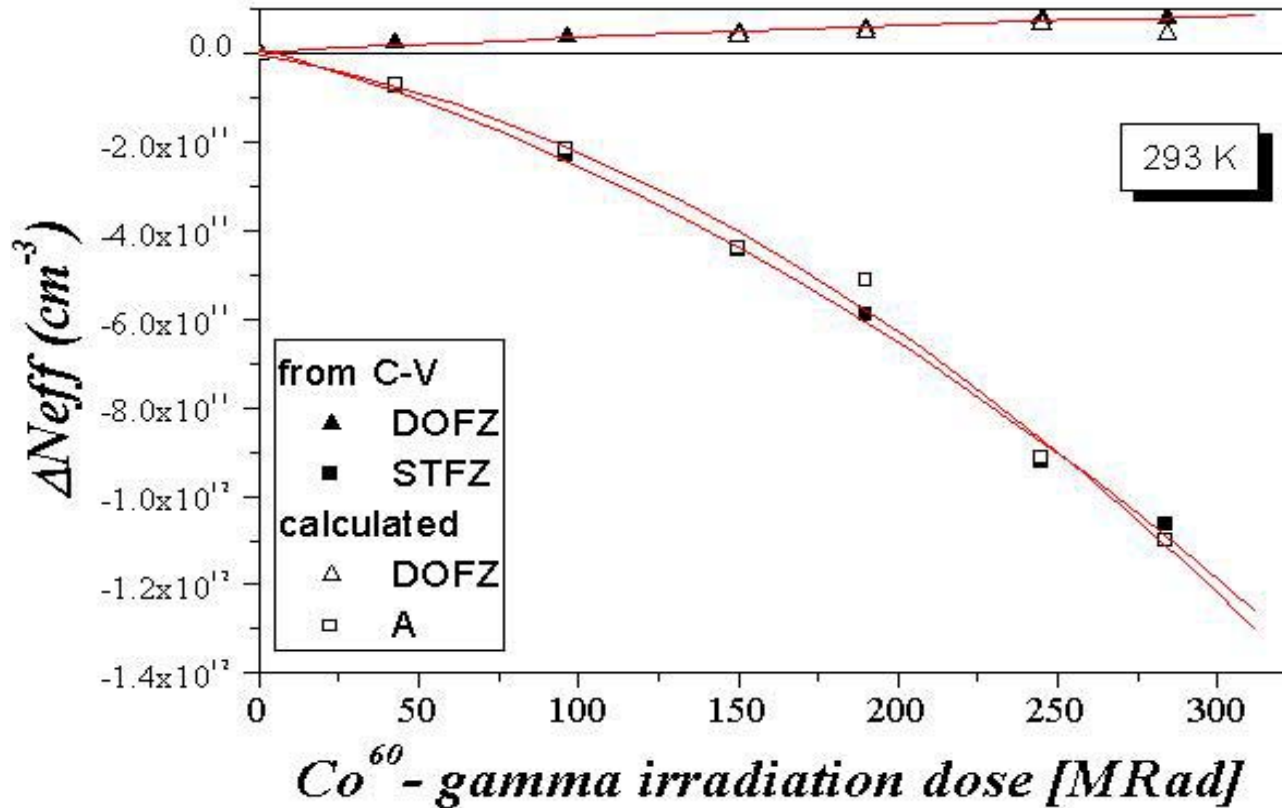
[Applied Physics Letters, 82, 2169, March 2003]

Levels responsible for macroscopic changes after γ -irradiation:

- **I-defect:** acceptor level at $E_C - 0.54\text{eV}$ (coming up for approx. 85% of damage)
peculiarity: quadratic dose dependence
⇒ good candidate for the V_2O defect
- **Γ -defect:** acceptor level at $E_V + 0.68\text{eV}$ (coming up for approx. 10% of damage)
- **BD-defect:** bistable shallow thermal donor (important in oxygen enriched silicon)



- Comparison for effective doping concentration for two different materials
 - as predicted by the microscopic measurements (open symbols)
 - as deduced from CV characteristics (filled symbols)



[I.Pintilie et al.,
NIMA514, 18, 2003]

- Excellent agreement also for the increase of leakage current (not shown here)

- DOFZ (Diffusion Oxygenated Float Zone Silicon)**

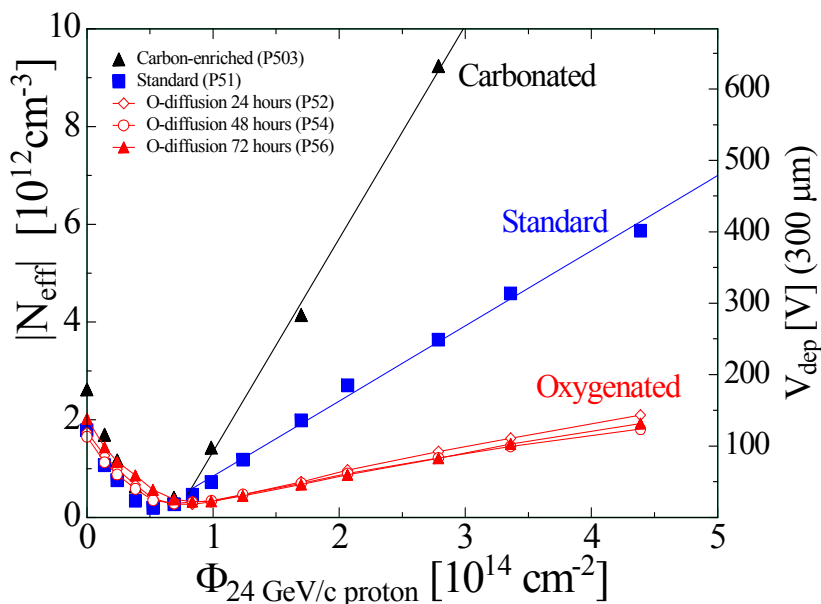
- 1982 First oxygen diffusion tests on FZ [Brotherton et al. J.Appl.Phys.,Vol.53, No.8.,5720]
- 1995 First tests on detector grade silicon [Z.Li et al. IEEE TNS Vol.42,No.4,219]
- **1999 Introduced to the HEP community by RD48 (ROSE)**

ROSE
RD48



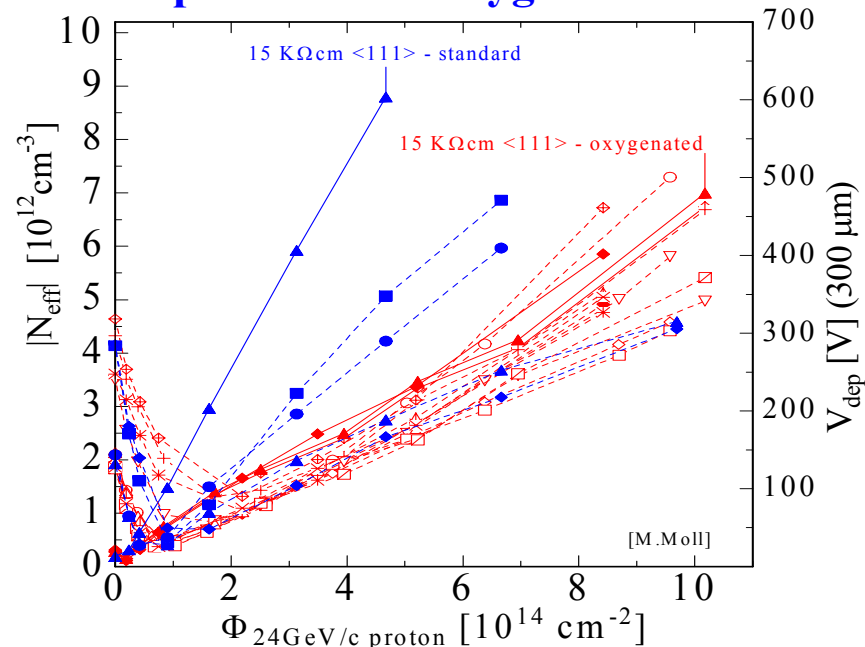
<http://cern.ch/rd48>

First tests in 1999 show clear advantage of oxygenation



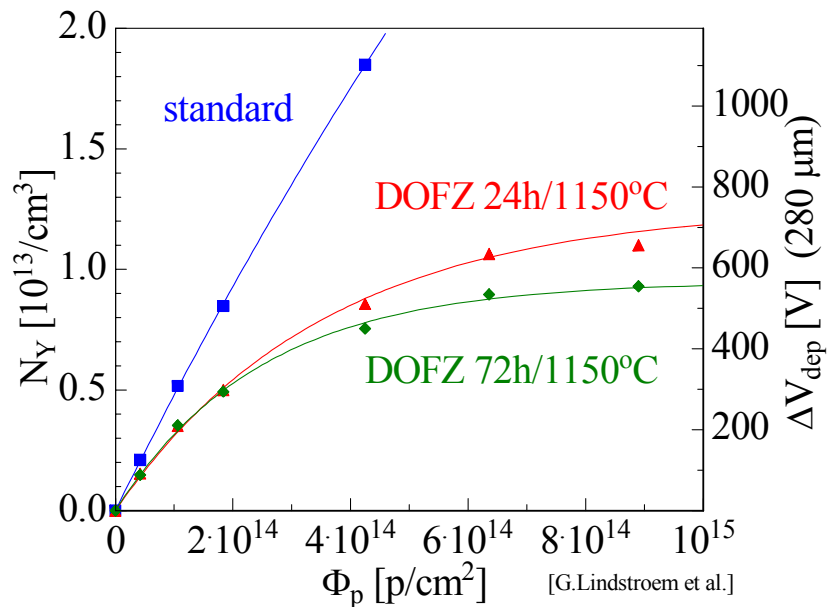
[RD48-NIMA 465(2001) 60]

Later systematic tests reveal strong variations with no clear dependence on oxygen content

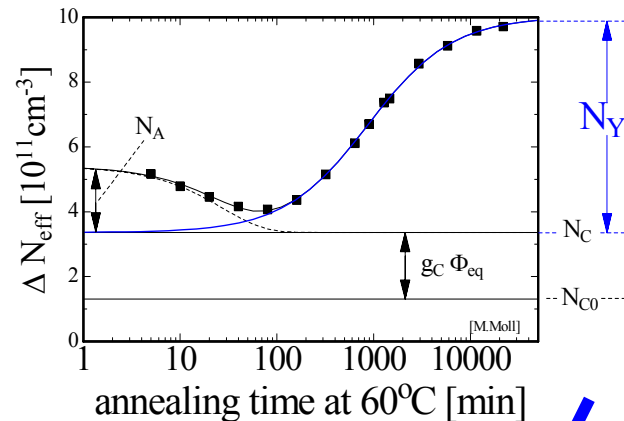


[RD50-NIMA 511 (2003) 97]

Reverse Annealing

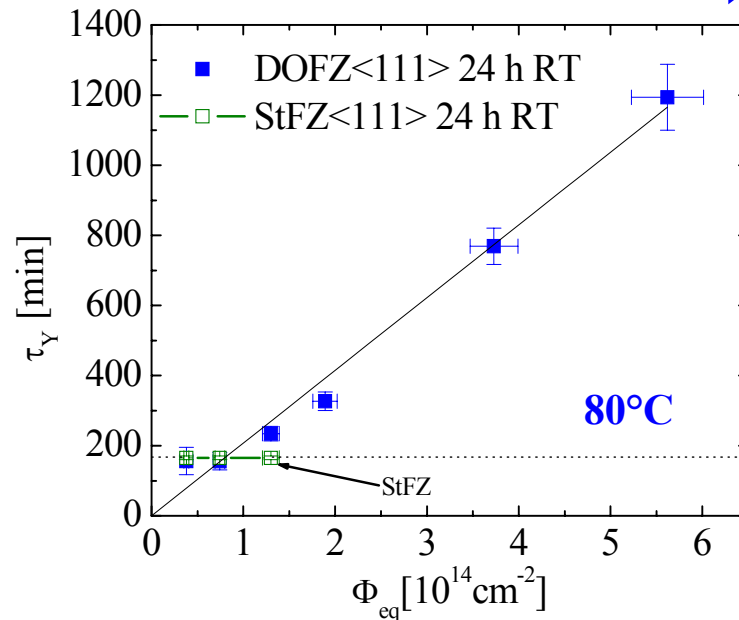


Saturation of amplitude

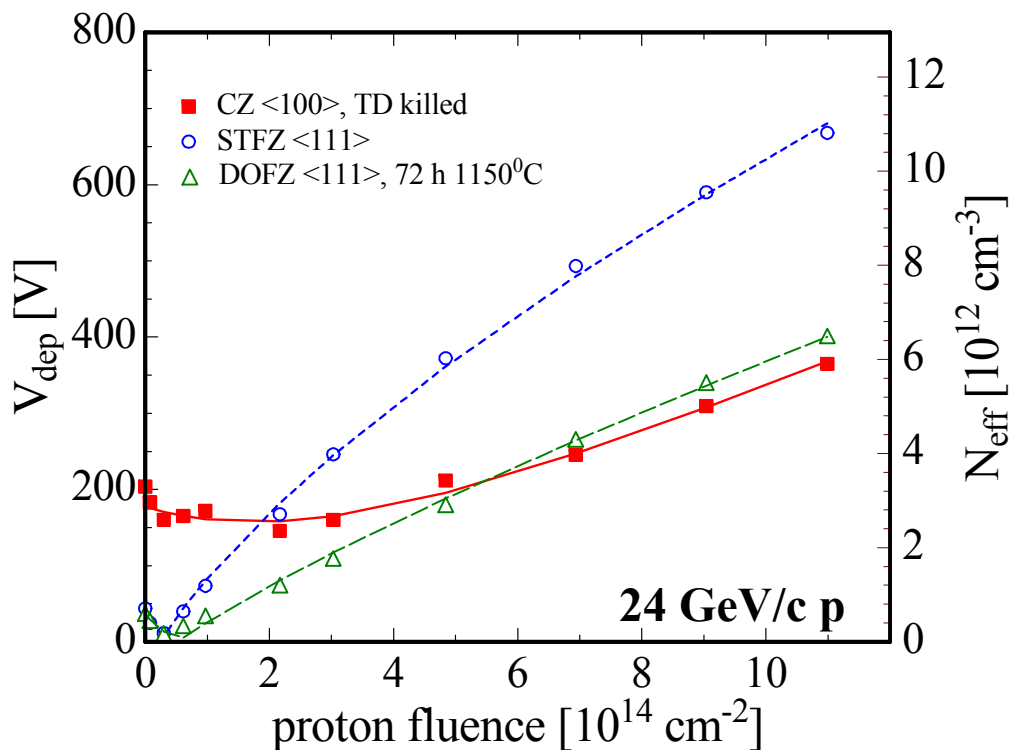


delayed reverse annealing

- **DOFZ: Saturation of reverse annealing** (24 GeV/c p - only little effect after neutron irradiation observed !)
- **DOFZ: No big difference between 24h and 72h oxidation at 1150°C**
- **DOFZ : time constant depending on fluence**



- **Very high Oxygen content 10^{17} - 10^{18}cm^{-3}** (Grown in quartz (SiO_2)crucible)
- **High resistivity ($>1\text{K}\Omega\text{cm}$) available only recently** (Magnetic CZ technology)

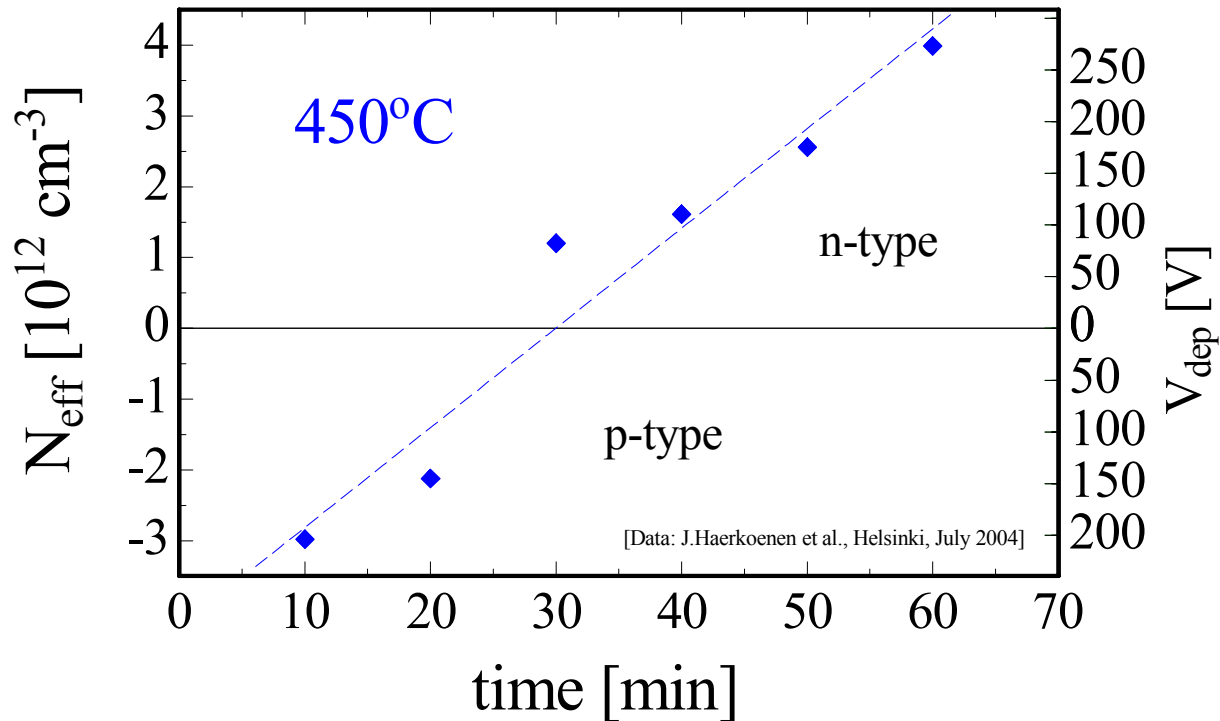


Irradiation of test-structures:

- **Only small change in V_{dep}**
 - $1 \cdot 10^{15}$ (190 MeV π)/ cm^2
 - $1 \cdot 10^{15}$ (24 GeV/c p)/ cm^2
 - $5 \cdot 10^{14}$ (10 MeV p)/ cm^2
- **No type inversion (Sumitomo CZ)**
(However, type inversion observed for Okmetic MCZ after $5 \cdot 10^{14}$ (10 MeV p)/ cm^2)
- **Leakage current and charge trapping as for FZ silicon**
- **Very high oxygen content:
Beware of thermal donors !**

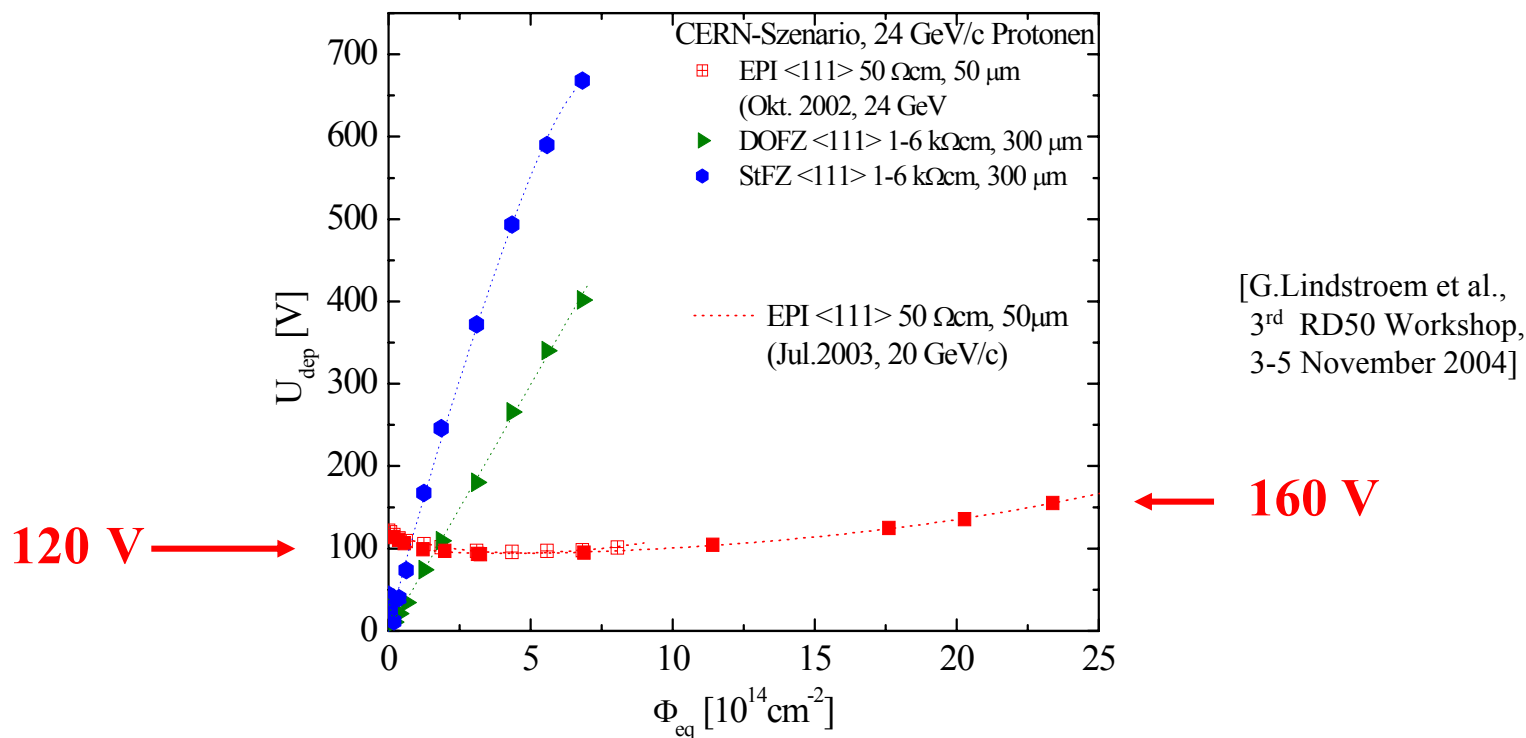
- Thermal Donor generation due to heat treatment at 450°C
- Effective doping concentration (depletion voltage) can be tailored

(here: starting with p-type material and converting it to n-type)



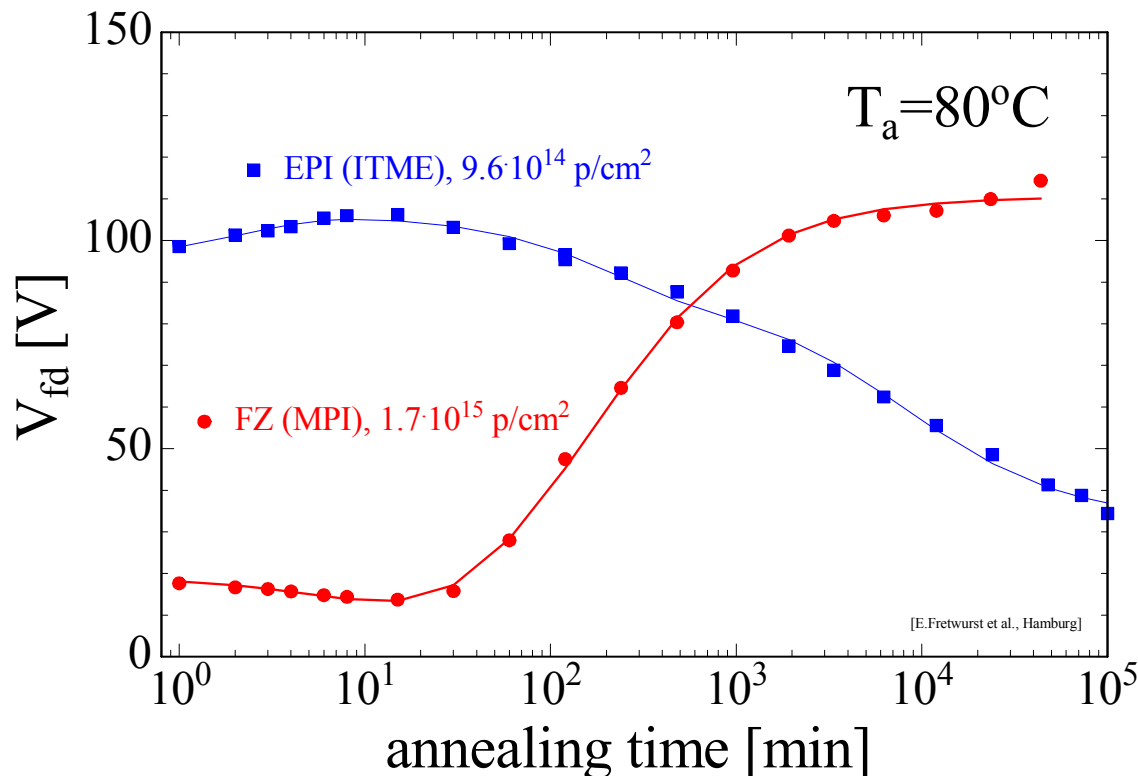
- Radiation hardness of thermal donor doped MCZ under test

- **Detectors: ITME epi-silicon** (50 μm , 50 Ωcm layer on CZ-substrate) **processed by CiS**
- **Irradiations:** 24 GeV/c protons, 58 MeV Li and reactor neutrons (up to $1 \cdot 10^{16} \text{cm}^{-2}$); **no type inversion** observed for proton irradiation



- **Leakage current almost identical to CZ, FZ, DOFZ detectors**

- 50 μm thick silicon detectors:
 - **Epitaxial silicon** (50 Ωcm on CZ substrate, ITME & CiS)
 - **Thin FZ silicon** (4K Ωcm , MPI Munich, wafer bonding technique)



[E.Fretwurst et al., 4th RD50 Workshop, 4-7 May 2004]

- **Thin FZ silicon:** Type inverted, increase of depletion voltage with time
- **Epitaxial silicon:** No type inversion, decrease of depletion voltage with time
 ⇒ No need for low temperature during maintenance of SLHC detectors!

Property	Diamond	GaN	4H SiC	Si
E_g [eV]	5.5	3.39	3.3	1.12
$E_{\text{breakdown}}$ [V/cm]	10^7	$4 \cdot 10^6$	$2.2 \cdot 10^6$	$3 \cdot 10^5$
μ_e [cm^2/Vs]	1800	1000	800	1450
μ_h [cm^2/Vs]	1200	30	115	450
v_{sat} [cm/s]	$2.2 \cdot 10^7$	-	$2 \cdot 10^7$	$0.8 \cdot 10^7$
Z	6	31/7	14/6	14
ϵ_r	5.7	9.6	9.7	11.9
e-h energy [eV]	13	8.9	7.6-8.4	3.6
Density [g/cm ³]	3.515	6.15	3.22	2.33
Displacem. [eV]	43	≥ 15	25	13-20

- Wide bandgap (3.3eV)
- ⇒ lower leakage current than silicon

- Signal:
- Diamond 36 e/ μm
- SiC 51 e/ μm
- Si 89 e/ μm

- ⇒ more charge than diamond

- Higher displacement threshold than silicon
- ⇒ radiation harder than silicon (?)

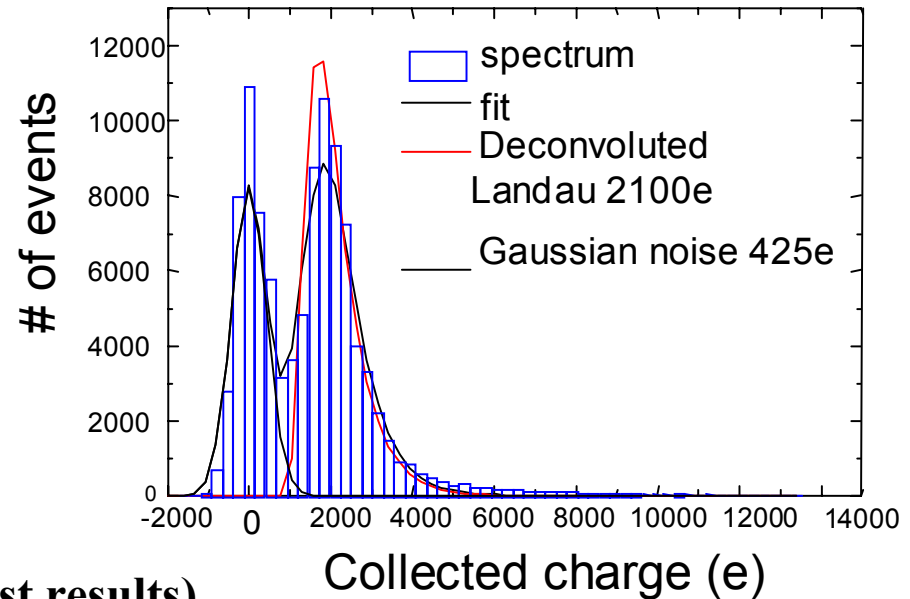
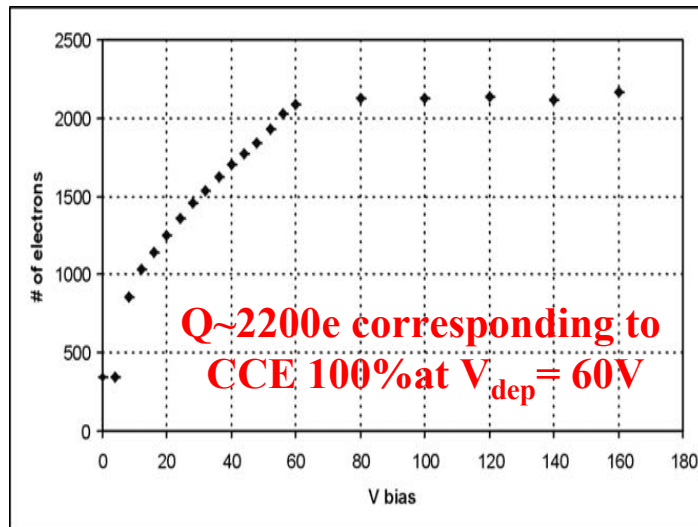
R&D on diamond detectors:
RD48 – Collaboration
<http://cern.ch/rd48/>
See talk of H.Kagan on Monday

- **Semi-Insulating SiC**

- $\rho > 10^{11} \Omega \text{cm}$ due to vanadium compensation
- CCE 60% in as-grown, $\sim 55\%$ after irradiation with 10^{13}cm^{-2} 300 MeV/c π
- Vanadium is responsible of incomplete charge collection

- **Epitaxial 4H-SiC**

- $N_{\text{eff}} \sim 5 \cdot 10^{13} \text{cm}^{-3}$; 40 μm by IKZ Berlin on CREE substrate (Schottky contacts)

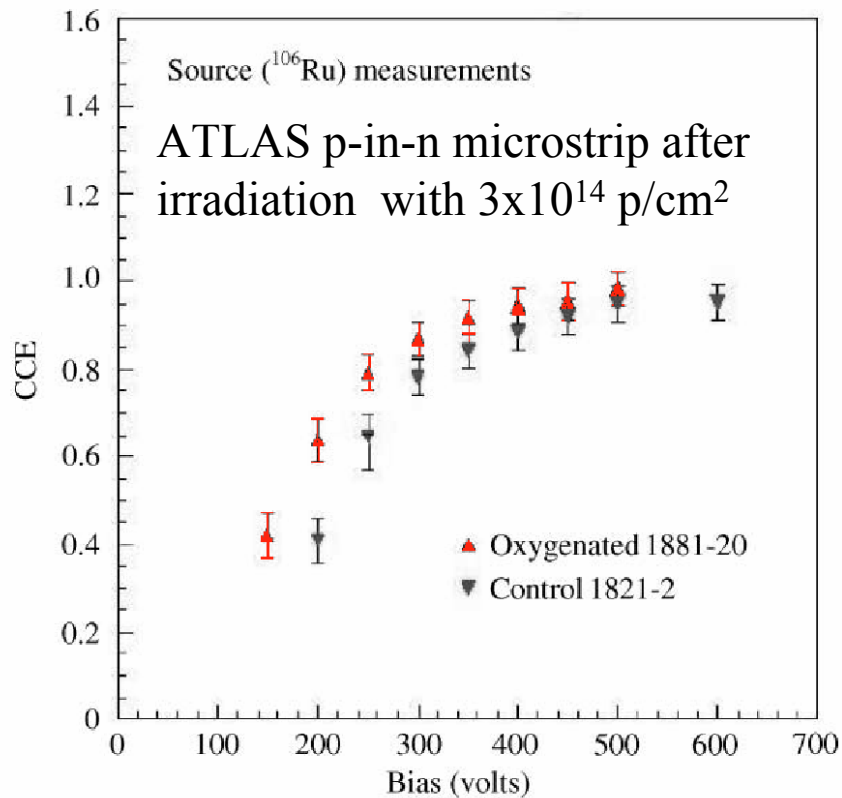


- **Irradiations: (Analysis in progress, first results)**

- deep levels identified with DLTS (0.18-1.22eV)
- CCE (alpha) going down to 80% after 10^{14}cm^{-2} 8MeV protons

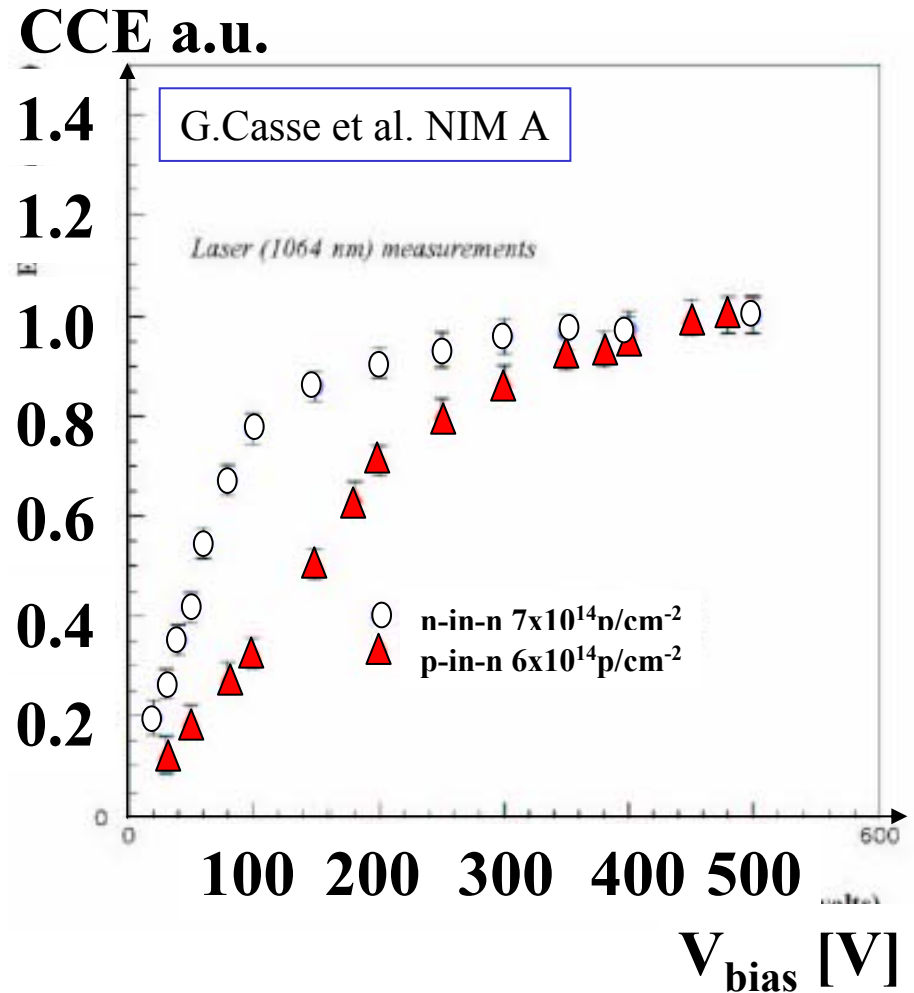
See following presentation by
James Grant (Glasgow)

- Beneficial effect of oxygen in p-irradiated DOFZ p-in-n microstrips almost disappears due to type inversion!

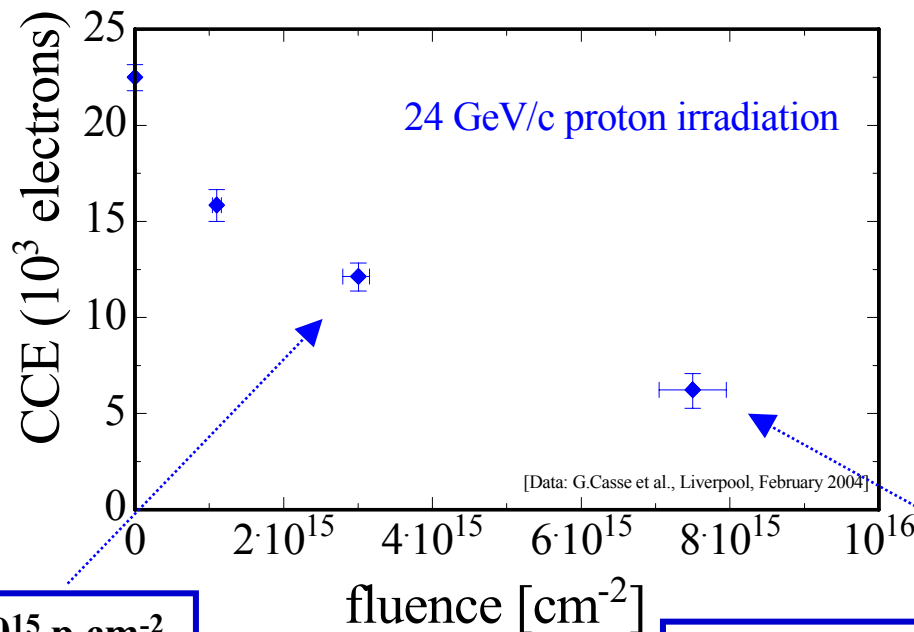


[G.Casse et al. NIM A 466 (2001) 335-344]

- Using n-in-n devices instead results in higher CCE.



- Miniature n-in-p microstrip detectors ($280\mu\text{m}$).
- Detectors read-out with LHC speed (40MHz) chip (SCT128A)
- Material: standard p-type and oxygenated (DOFZ) p-type
- Irradiation:



G. Casse et al., Feb. 2004

CCE $\sim 60\%$ after $3 \cdot 10^{15} \text{ p cm}^{-2}$
at 900V (standard p-type)

CCE $\sim 30\%$ after $7.5 \cdot 10^{15} \text{ p cm}^{-2}$
900V (oxygenated p-type)

At the highest fluence $Q \sim 6500e$ at $V_{\text{bias}} = 900\text{V}$ corresponding to: $\text{ccd} \sim 90\mu\text{m}$

Motivation: After 1 MeV neutron irradiation to 10^{15} cm^{-2}
the effective drift length for e is $\sim 150 \mu\text{m}$ and for h $\sim 50 \mu\text{m}$

\Rightarrow use thin detectors (50-100 μm) from the beginning

• **Benefits:**

- low operating voltage
- improved radiation tolerance: - 50 μm thick, 50 Ωcm Si detector ($V_{\text{dep}} = 200\text{V}$):
- type inversion only after 10^{15} cm^{-2}

• **Drawbacks:**

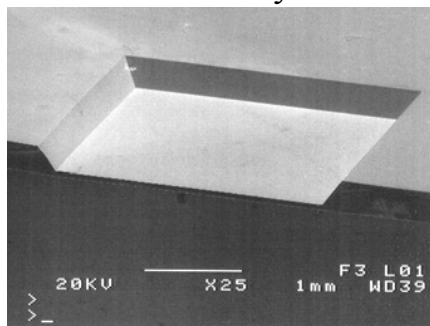
- mip signal $\sim 3500\text{e-h}$ pairs

• **Technical Approaches:**

- Epitaxial Si device (shown before)
- Thinning with chemical attacks (IRST, Italy)
and wafer bonding technology (MPI Munich, Germany)

See following presentation by
Marco Petasecca on thin Si detectors

IRST: SEM of a silicon wafer thinned by TMAH



Cross section of a thinned silicon detector



Devices produced end of 2003 (up to 11mm²):

Thickness [μm]	Leakage Current [nA/cm^3]	V_{dep} [V]
300	80	12
99	30	~ 1
57	55	< 1

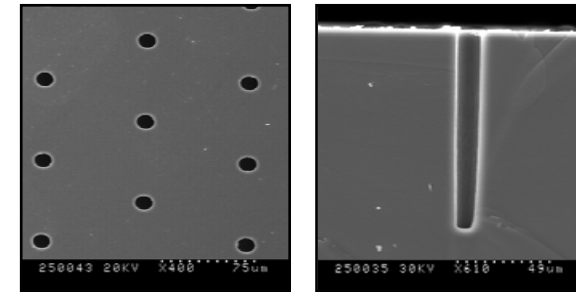
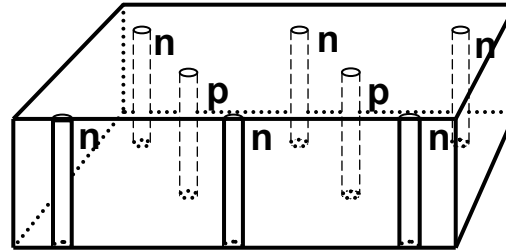
Electrodes:

- narrow columns along detector thickness-“3D”
- diameter: $10\mu\text{m}$ distance: $50 - 100\mu\text{m}$

See overview talk by:
Sherwood Parker (Monday)

Lateral depletion:

- lower depletion voltage needed
- thicker detectors possible
- fast signal



Glasgow University

Hole processing :

- Dry etching, Laser drilling, Photo Electro Chemical
- Present aspect ratio (RD50) 13:1, Target: 30:1

Electrode material

- Doped Polysilicon (Si)
- Schottky (GaAs)

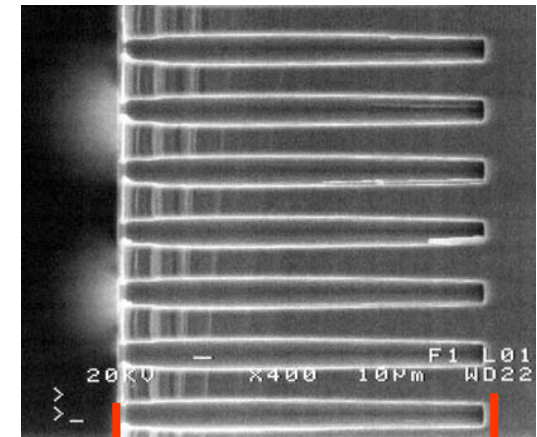
3D detector developments within RD50:

1) Glasgow University – Schottky contacts
(see Poster of Victoria Wright)

2) IRST-Trento and CNM Barcelona (since 2003)

CNM: Hole etching (DRIE); IRST: all further processing
diffused contacts or doped polysilicon deposition

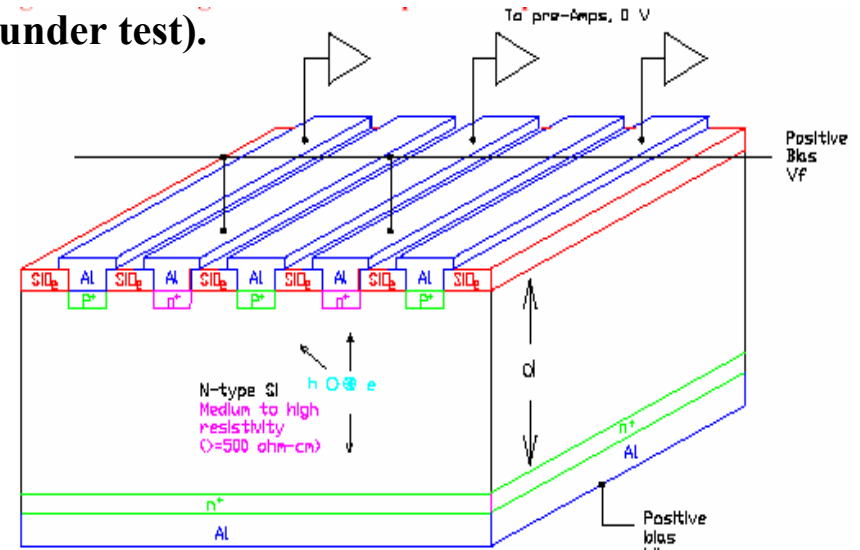
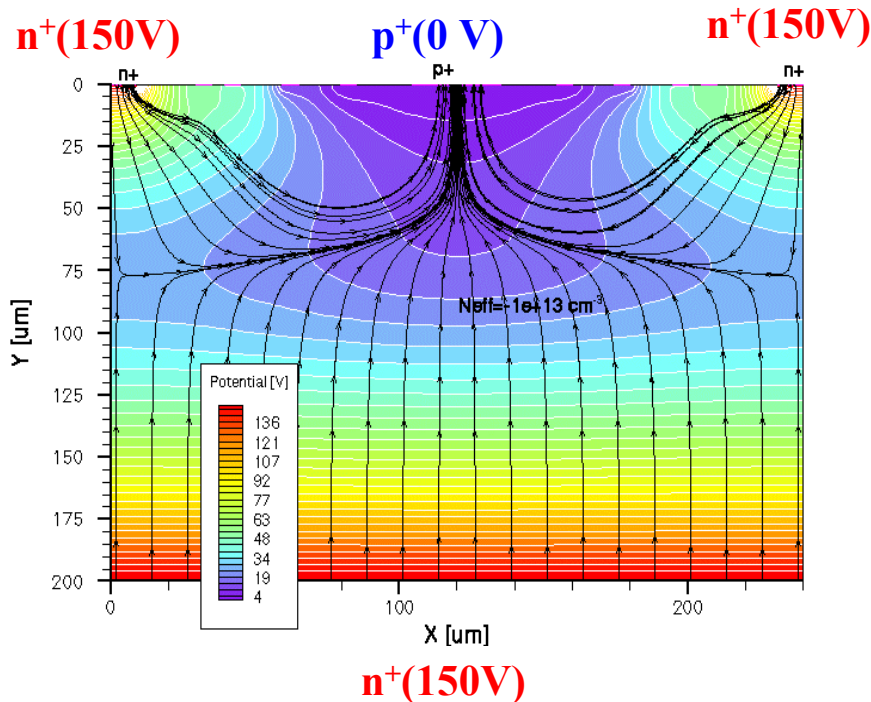
hole diameter $15\mu\text{m}$



~200 micron

Semi 3-D devices proposed by Z. Li, BNL.

- Planar technology easier to process than 3D sensors
- Single-sided processing
- Large reduction in detector full depletion voltage after type inversion
- Processing of first prototype completed (presently under test).



Z. Li et al. NIMA478, (2002), 303-310



Simulation of electric profile in semi 3D after irradiation to $5 \times 10^{14} \text{ n/cm}^2$.

- Radiation hard materials and new device concepts for Super LHC tracking detectors are under study by the RD50 collaboration.
- At fluences up to 10^{15}cm^{-2} (Outer layers of a SLHC detector) the change of the depletion voltage and the large area to be covered by detectors is the major problem. **CZ detectors could be a cost-effective radiation hard solution.**
- Miniature microstrip and pixel detectors made with defect engineered Si have been fabricated by RD50 or are in process. First tests with LHC like electronics are encouraging: CCE on microstrip n-in-p oxygenated detectors irradiated up to 7×10^{15} [$24 \text{ GeV}/c \text{ p}/\text{cm}^2$] is $> 6500 \text{ e}$.
- At the fluence of 10^{16}cm^{-2} (Innermost layer of a SLHC detector) the active thickness of any silicon material is significantly reduced due to trapping. The two most promising options so far are:
 - **Thin/EPI detectors : drawback: radiation hard electronics for low signals needed**
 - **3D detectors : drawback: technology has to be optimized**
- **New Materials** like SiC and GaN have been characterized. However, thicker samples and more radiation studies are needed to assess if these materials could be an alternative to Silicon.

Further information: <http://cern.ch/rd50/>